

1 **Crystallization kinetics of clinopyroxene and titanomagnetite growing from a trachybasaltic**
2 **melt: New insights from isothermal time-series experiments**

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35 **Abstract**

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In order to investigate the role of crystallization kinetics in mafic alkaline systems, textural measurements, mineral compositional changes and diffusion modelling calculations have been carried out on isothermal time-series experiments. The data were obtained at 400 MPa and 1,100 °C under anhydrous (nominally 0 wt.% H₂O) and hydrous (2 wt.% H₂O added) conditions. A synthetic trachybasaltic melt was first heated up to the superliquidus temperature of 1,300 °C and then rapidly cooled at 80 °C/min down to 1,100 °C. The final target temperature was kept constant over variable dwell times in the range of 0.5-24 h. Estimates of area fractions, crystal sizes, crystal size distributions and surface area to volume ratios indicate the attainment of fast crystal growth kinetics at the shortest experimental run duration, with early achievement of stable crystal sizes for clinopyroxene and titanomagnetite. The surface area to volume ratio weakly decreases with increasing dwell time, according to the development of euhedral crystal morphologies. Crystal growth rates are also observed to progressively decrease from 0.5 to 24 h. Due to the effect of fast growth kinetics, the morphological maturation of clinopyroxene progresses by attachment of dendrite branches, infilling and overgrowth phenomena, leading to the formation of well-faced and euhedral crystals. The kinetically-controlled cation exchange (Si + Mg) → (T^{Al} + Fe³⁺) controls the clinopyroxene compositional variation, expanding the stability of Tschermak component at the expense of diopside. Conversely, titanomagnetite is characterized by an almost constant composition that, however, is enriched in incompatible Al and Mg cations, as typically observed under rapid crystal growth conditions. Titanomagnetite crystals show always euhedral morphology that develops by heterogeneous nucleation on early-formed clinopyroxene dendrites. With increasing dwell time, the textural maturation of clinopyroxene reduces the number of heterogeneous nucleation sites and the titanomagnetite growth proceeds by coarsening. Overall, the effect of undercooling causes strong supersaturation phenomena in the trachybasaltic melt, resulting in enhanced nucleation kinetics and fast attainment of a high crystallinity. As the dwell time increases, the bulk system tends to minimize the interfacial free energy between crystals and surrounding melt. This results in the progressive replacement of the early dendritic shapes developed in a diffusion-limited growth regime, by the formation of euhedral morphologies typical of interface-limited regimes that still retain the chemical evidences of the dendritic stage as complex zoning patterns in clinopyroxene.

66 **Keywords:** Crystallization kinetics; undercooling; clinopyroxene; titanomagnetite; textural
67 analysis; rapid crystal growth
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69 1. Introduction

70 The textural features of igneous rocks represent a valuable source of information for better
71 understanding the physicochemical conditions of magmas at which crystals nucleate and grow (e.g.
72 Cashman 1990, 1993; Armienti et al., 1994, 2013; Cashman and Blundy, 2000; Hammer, 2006,
73 2008; Resmini, 2007; Armienti, 2008). In this context, crystallization kinetics exert an important
74 role in the solidification of the melt into a rock (e.g. Cashman and Marsh, 1988; Marsh, 1988, 1998;
75 Hersum and Marsh, 2006; Higgins, 1998, 2006; Zieg and Lofgren, 2006; Spillar and Dolejs, 2013;
76 Mollo and Hammer, 2017). A great number of experimental studies has been conducted on either
77 natural or synthetic silicate liquids of variable compositions, in order to provide constraints on the
78 textural and chemical changes of minerals under the effects of variable degrees of undercooling
79 caused by either cooling (e.g., Lofgren et al., 1974; Donaldson, 1976; Walker et al., 1976, 1978;
80 Kirkpatrick, 1981; Lofgren and Russell, 1986; Pupier et al., 2008; Del Gaudio et al., 2010; Mollo et
81 al., 2011, 2012a, 2013a, 2013b; Iezzi et al., 2013) or decompression (e.g., Hammer and Rutherford,
82 2002, Couch et al., 2003; Hammer, 2008; Brugger and Hammer, 2010a; Waters et al., 2015).
83 Undercooling, expressed as the difference between the liquidus and experimental temperatures (ΔT
84 = $T_{\text{liq}} - T_{\text{exp}}$), can be experimentally controlled by either varying the liquidus temperature (by
85 changing the melt H₂O content) or the final target temperature (by applying variable cooling rates)
86 (cf. Hammer, 2008; Mollo & Hammer, 2017). Ostwald ripening, crystal coalescence, grain
87 boundary migration, and infilling of early crystal framework are primary controlled by the
88 undercooling and documented as the effective mechanisms controlling the textural features of the
89 experimental products (Hammer and Rutherford, 2002; Simakin et al., 2003, Zieg and Lofgren,
90 2006; Pupier et al., 2008; Schiavi et al., 2009; Mollo et al., 2010; Iezzi et al., 2011, 2014; Ni et al.,
91 2014; Welsch et al., 2014; Shea et al., 2015). The significance of these textural maturation
92 mechanisms has been also assessed through specifically designed laboratory (Park and Hanson,
93 1999; Cabane et al., 2001, 2005) and natural (Higgins, 1998, 1999; O’Driscoll et al., 2007; Higgins
94 and Roberge, 2003; Mollo et al., 2015a) investigations, as well as through experimental analogs
95 (Means and Park, 1994) and numerical modelling (Hersum and Marsh, 2006). However, in spite of
96 this extensive literature, a few experimental studies have clearly addressed the role of relaxation
97 kinetics on crystal growth when the resting temperature is kept constant over different dwell times,
98 after an early stage of undercooling (cf. Mollo et al., 2012a).

99 In this study, we present new textural and chemical data on clinopyroxene and
100 titanomagnetite crystals obtained through isothermal time-series experiments conducted on a
101 synthetic trachybasalt rapidly undercooled before crystallization. The imposed degree of
102 undercooling is the thermodynamic driving force inducing early nucleation of clinopyroxene and

103 titanomagnetite, whilst the crystal growth and textural maturation proceeded isothermally over time.
104 The effects of crystallization kinetics and possible equilibration phenomena with increasing dwell
105 time are quantified in terms of mineral and glass compositions, total crystallinity, maximum crystal
106 size, and crystal size distribution analysis. Textural data are also compared with the diffusion length
107 of chemical elements in the melt (i.e., modeling data from [Zhang et al., 2010](#)), in order to
108 discriminate the influence of diffusion- and interface-limited growth on the disequilibrium or near-
109 equilibrium compositions of minerals ([Lofgren et al., 2006](#); [Hammer, 2008](#); [Watson & Muller,
110 2009](#); [Mollo and Hammer, 2017](#)), and the potential relationship with observed crystal morphologies
111 ([Sunagawa, 1981, 2005](#); [Faure et al., 2007](#)). This comparative approach provides new insights on
112 the solidification behavior of mafic alkaline magmas.

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2. Methods

2.1. Experiments

117 The time-series experiments were conducted at the HP-HT Laboratory of Experimental
118 Volcanology and Geophysics of the Istituto Nazionale di Geofisica e Vulcanologia (INGV) in
119 Rome, Italy. A synthetic trachybasaltic starting material was obtained in batches of ~2 gr from pure
120 oxides and carbonates mixed by grinding under ethanol in an agate mortar for ~1 h. This mixture
121 was chosen in order to reproduce one of the most primitive products erupted at Mt. Etna volcano
122 (i.e., the Mt. Maletto formation in Sicily, Italy; [Armienti et al., 1988, 2013](#); [Mollo et al., 2015b](#)), as
123 well as mafic alkaline magmas from different and complex intraplate settings in the world (cf.
124 [Mollo et al., 2018](#)). A Pt-crucible containing the synthetic powder was loaded in a 1 atm vertical
125 tube CO–CO₂ gas-mixing furnace at the NNO+2 buffer. The temperature was kept at 1,100 °C for
126 10 h to ensure decarbonation and then was raised up to 1,600 °C. This temperature was kept
127 constant for 4 h to ensure homogeneous melting. The resulting glass was removed from the Pt-
128 crucible and powdered. Backscattered microphotographs and microprobe analyses performed on
129 chips extracted from top, middle, and bottom of the Pt-crucible, demonstrated homogeneity and the
130 absence of crystalline phases in a glass with an average composition (in wt.%) of SiO₂ = 50.09
131 (±0.29), TiO₂ = 1.49 (±0.1), Al₂O₃ = 15.92 (±0.18), FeO = 8.09 (±0.13), MgO = 7.60 (±0.16), CaO
132 = 11.97 (±0.20), Na₂O = 3.26 (±0.12), K₂O = 1.53 (±0.09), and P₂O₅ = 0.02 (±0.02). In order to
133 minimize iron loss, some aliquots of the powder were previously loaded into the Pt-crucible and run
134 for 3 h at 1,600 °C to pre-saturate the crucible (cf. [Conte et al., 2006](#)). The sample holder was then
135 quenched and cleaned in a hot HF solution. The same approach was adopted to pre-saturate the Pt-

136 capsules (3 mm of outer diameter) used for the piston cylinder experiments conducted at 400 MPa.
137 It is found that the iron loss from the samples was kept to <5% of the initial amount.

138 Piston cylinder experiments were carried out with a non-end loaded apparatus
139 (“QUICKpress”, Depths of the Earth co.) using a 19-mm NaCl-pyrex-graphite-MgO assembly that
140 produced a redox state close to the NNO+2 buffer (Masotta et al., 2012). The assembly was
141 simultaneously loaded with two Pt-capsules containing the nominally anhydrous (dried in oven at
142 110 °C for 48 h) and hydrous (2 wt.% of deionized H₂O added with a microsyringe) trachybasaltic
143 glass. The capsules were also surrounded by powdered pyrophyllite to prevent H₂O loss and
144 enhance stress homogenization during initial compression (Freda et al. 2001, 2008). After cold
145 pressurization to a nominal pressure 10% higher than desired, the pressure was decreased down to
146 400 MPa, after reaching the final resting temperature. Reversal experiments were carried out by
147 isobarically superheating the starting glass from room temperature up to the superliquidus condition
148 of 1,300 °C (the liquidus temperature of the trachybasalt is ~1,200 °C; Del Gaudio et al., 2010;
149 Mollo et al., 2010, 2011) at a rate of 80 °C/min. This temperature was kept constant for 30 min and
150 then decreased down to the target crystallization condition of 1,100 °C using the same rate of 80
151 °C/min. At the superliquidus temperature of 1,300 °C, the relaxation time to reach equilibrium is
152 much shorter than the 30 min for the used trachybasaltic melt (Webb, 1997). For this starting melt
153 composition and the used experimental strategy, relaxation kinetics are extremely rapid in time (i.e.,
154 from milli- to micro-seconds) and weakly dependent on the superheating path used in laboratory
155 (Vetere et al., 2013, 2015). The temperature was monitored by a factory-calibrated C-type (W-
156 5Re/W-26Re) thermocouple with precision of ±3 °C. Considering the limited effect of 2 wt.% H₂O
157 on the liquidus region of the trachybasaltic melt, the cooling path produced a nominal undercooling
158 condition relatively fast ($\Delta T \approx 100$ -120 °C), as previously experimentally demonstrated for similar
159 starting compositions (Mollo et al., 2013a, 2013b, 2013c). ΔT refers to the difference between the
160 phase-in temperature of the melt and the quench temperature. The isothermal condition was kept
161 constant over variable dwell times of 0.5, 1, 2, 4, 8, and 24 h (Table 1). Then, an isobaric quench of
162 100 °C/s was applied.

163 164 2.2. Analyses

165 Microchemical analyses (Table 1S, 2S and 3S) were performed at the HP-HT Lab of INGV
166 using an electron probe microanalyzer (EPMA) Jeol-JXA8200 with combined EDS-WDS (five
167 spectrometers with twelve crystals). Data were collected using 15 kV accelerating voltage and 10
168 nA beam current. For glasses, a slightly defocused electron beam with a size of 5 μm was used,
169 with a counting time of 5 s on background and 15 s on peak. For crystals, the beam size was 5 μm

170 with a counting time of 20 and 10 s on peaks and background respectively. The following standards
171 were used: jadeite (Si and Na), corundum (Al), forsterite (Mg), andradite (Fe), rutile (Ti), orthoclase
172 (K), apatite (P), and spessartine (Mn). Sodium and potassium were analyzed first to minimize alkali
173 migration effects. The precision of the microprobe was measured through the analysis of well-
174 characterized synthetic oxide and mineral secondary standards. Based on counting statistics,
175 analytical uncertainties relative to their reported concentrations indicate that precision was better
176 than 5% for all cations.

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178 *2.3. Image processing and textural parameters*

179 Microphotographs were collected in backscattered electron (BSE) mode of a field emission
180 gun-scanning electron microscopy (FE-SEM) Jeol 6500F equipped with an energy-dispersive
181 spectrometer (EDS) detector and installed at the HP-HT Lab of INGV. Magnifications from 50× to
182 400× were adopted to collect a statistically representative number of crystals (i.e., clinopyroxene
183 and titanomagnetite) for each different experimental charge. The ImageJ software package was used
184 for image processing. Textural data were determined by thresholding (i.e., segmentation, as showed
185 in [Fig.1](#)), counting, and measuring the crystal edges (e.g. [Hammer et al., 1999](#); [Brugger and
186 Hammer, 2010a](#)). According to the stereological theorem of [Delesse \(1847\)](#), the area fraction of the
187 minerals investigated equals their volume fraction, provided that the distribution of crystals remains
188 uniform along the entire length of the experimental capsule. Boundaries between touching crystals
189 were identified through the visual inspection of the crystal shapes. The uncertainty introduced by
190 this refining process was considered negligible due to the statistically high number of particles
191 examined for each experimental capsule: 266–1,277 and 376–1,914 crystals for clinopyroxene and
192 titanomagnetite, respectively. The uncertainty related to the quantification of the textural parameters
193 was calculated as 1σ variation of different crystal populations ([Tables 2 and 3](#)), thus representing
194 sample heterogeneity rather than the error caused by image processing. A different approach is
195 employed for estimating the uncertainty associated with area fraction (%) measurements. It is
196 apparent that the segmentation process represents the main potential source of error in estimating
197 area fraction, especially for crystals with shapes characterized by high surface-to-volume ratios (e.g.
198 dendritic crystals). Therefore, for each sample, we considered positive and negative errors in area
199 fraction estimates that would arise from addition and subtraction of a pixel layer around each crystal
200 in the binary image.

201 In the case of clinopyroxene, dendritic crystals characterize most of the experimental
202 charges and their multiple intersection with the section plane impeded any reliable crystal size
203 distribution (CSD) analysis ([Hammer, 2006](#); [Higgins, 2006](#); [Armienti, 2008](#)). Conversely, other

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204 textural parameters can be effectively quantified in terms of the dimensions of largest crystals (i.e.,
205 best-fit ellipse major and minor axis; Fig. 2), the surface area to volume ratio, and the volume
206 fraction. The maximum growth rate (G_{max}) of clinopyroxene was measured following one of the
207 most common methods reported in literature (Burkhard, 2002; Hammer and Rutherford, 2002;
208 Couch et al., 2003; Baker, 2008; Iezzi et al., 2011):

$$209 \quad G_{max} = (L W)^{0.5} / (2 t) \quad (1)$$

210 Where L and W are the mean length and width, respectively, on the basis of the ten largest crystals
211 for each BSE image, whereas t is the dwell time. The factor 2 refers to the growth of half-crystal
212 during advancement of the pinacoid face. To track the textural evolution of clinopyroxene as a
213 function of dwell time, the surface area to volume ratio (S_v^P) has been also determined according to
214 the method outlined by Hammer (2006, 2009):

$$215 \quad S_v^P = S_v / \phi \quad (2)$$

216 and

$$217 \quad S_v = 2 N_L \quad (3)$$

218 Where S_v and ϕ are the total interfacial area of the population per unit volume of sample and the
219 fractional volumetric phase abundance. N_L is the mean number of boundary intersections per unit
220 length of randomly-oriented test line (cf. Underwood, 1968). The parameter S_v^P represents the ratio
221 of crystal population surface area to the volume of that population, thus accounting for the
222 difference in volumetric abundances among distinct crystal populations and/or BSE images. This
223 parameter allows to quantify the textural maturation based on the crystal morphology. Moreover,
224 S_v^P can be hypothetically related to the crystallization conditions of the system (i.e., diffusion- and
225 interface-limited crystallization phenomena), as long as it represents an estimate of the interfacial
226 area developing between the crystal surface and surrounding glass feeding the crystal growth
227 (Lofgren, 1974; Hammer and Rutherford, 2002).

228 In the case of titanomagnetite, the majority of crystals exhibits well-developed, euhedral
229 morphologies enabling reliable CSD analysis. The crystal growth was investigated through the
230 application of three different methods based on the assumption of constant growth kinetics over the
231 adopted dwell time (e.g., Marsh, 1988; Armienti, 2008). In Method 1, G_{max} was determined

238 following the same approach used for clinopyroxene. In Method 2, the batch growth rate (G) was
 239 determined through the expressions (Blundy and Cashman, 2008; Brugger and Hammer, 2010a,
 240 2010b):

$$241 \quad G = d / 2 t \quad (4)$$

242 and

$$243 \quad d = \sqrt{\frac{\phi}{N_A}} \quad (5)$$

244 and

$$245 \quad N_V = N_A / d \quad (6)$$

246 Where d , t , ϕ , N_A , and N_V are, respectively, the characteristic crystal size, the experimental time, the
 247 area fraction, the area number density (i.e., total number of crystals per unit area), and the
 248 volumetric number density. In Method 3, the CSD analysis was performed according to the
 249 procedure reported in Armienti (2008), where the stereological correction accounts for the
 250 “unfolding” method based on the algorithms of Schwartz (1939) and Saltykov (1949). This allows
 251 to consider the cut-section effects (i.e., larger particles contribute to smaller size classes when
 252 sectioned by a plane not passing through their center) and intersection-probability effects (i.e.,
 253 smaller particles are less likely intersected by the section plane; Higgins, 2000, 2006). Despite the
 254 shape of CSD curves is strongly dependent on the choice of the bin size, the procedure of Armienti
 255 (2008) is optimized by including a routine properly designed for the minimization of the residuals
 256 between the measured number of particles and that re-calculated from the CSD (zeroth moment of
 257 the distribution) analysis as follow:

$$258 \quad N_{tot} = Area \sum L_i N_V(L_i) \quad (7)$$

259 and between the measured volume crystal fraction and that resulting from the CSD (third moment
 260 of the distribution):

$$261 \quad V_f = \sum N_V(L_i) \frac{4}{3} \pi \left(\frac{L_i}{2}\right)^3 = \frac{\pi}{6} \sum L_i^3 N_V(L_i) \quad (8)$$

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Subsequently, the CSD plot is formulated for a given image and the slope of this distribution is determined. The growth rate and the characteristic crystal size were calculated as the ratio between the total length and total number of crystals, expressed as the integral forms of the first and zeroth moments of the distribution, respectively (Randolph and Larson, 1971; Cashman, 1992; Marsh, 1998, 2007):

$$L_d = \frac{L_{tot}}{N_{tot}} = \frac{\int_0^{\infty} L n^0 Gt dL}{\int_0^{\infty} n^0 Gt dL} = \frac{n^0 (Gt)^2}{n^0 Gt} = Gt \quad (9)$$

and

$$Slope = -\frac{1}{Gt} = -\frac{1}{L_d} \quad (10)$$

Where n^0 represents the value of the intercept at $L = 0$ and L_d is the characteristic crystal size. Note that L_d is different from d that, in turn, is the characteristic crystal size determined by the “batch” calculation approach.

3. Results

3.1. Textural data

3.1.1. Crystal morphologies

Clinopyroxene is the dominant mineral phase in the whole set of experiments (Fig. 3), with area fraction ranging from 35.8% to 53.5% (Table 2 and Fig. 4a). The comparison between anhydrous and hydrous experiments does not show relevant differences in the clinopyroxene content, neither illustrates any significant nor systematic variation with dwell time. Indeed, most of the area fraction measurements overlap within their associated uncertainties (Table 2 and Fig. 4a). Crystal morphologies are prevalently dendritic from 0.5 to 8 h, whereas well-developed textures are observed at 24 h (Figs. 2 and 3). A marked difference in the crystal habit is recognizable between anhydrous and hydrous conditions, with more euhedral and faceted clinopyroxenes found in hydrous experiments (Fig. 3). The mean maximum size of clinopyroxene ranges from 54 μm (0.5 h) to 86 μm (24h) and from 63 μm (0.5 h) to 120 μm (24h) in the anhydrous and hydrous charges, respectively (Fig. 4b). A significant increase in the crystal size occurs only at 24 h and this is particularly evident under hydrous conditions. The S_v^P parameter does not substantially change at

304 0.5, 1, and 2 h, but then starts to decrease as the dwell time increases (Fig. 5). Values of S_v^P
305 measured under hydrous conditions are, on average, lower than those measured for the anhydrous
306 charges. A marked difference can be observed at 24 h (Fig. 5), where the lower S_v^P value of hydrous
307 crystallization experiments reflects a higher degree of crystal euhedrality.

308 Titanomagnetite is scarce in the experimental charges, with area fractions variable from
309 0.5% to 2.5% and from 1.4% to 2.4% under anhydrous and hydrous crystallization conditions,
310 respectively (Table 3 and Fig.4c). At 0.5 and 1 h, the amount (0.5-0.8%) of crystals from anhydrous
311 experiments is apparently lower than that (1.4-1.8%) measured in the hydrous ones, despite area
312 fraction estimates are affected by great uncertainties (Fig.4c). Moreover, no clear trends are
313 observed with increasing dwell time (Fig.4c). The titanomagnetite morphology is well-developed
314 with most of the crystals showing euhedral shapes and only rare subhedral (hollow-to-hopper)
315 textures. Highly dendritic morphologies, as those obtained in fast (up to 15 °C/min) cooling rate
316 experiments (Hammer, 2006; Mollo et al., 2013b) are never observed. The mean maximum crystal
317 size varies from 5 μm (0.5 h) to 11 μm (24 h) and from 7 μm (0.5 h) to 23 μm (24 h) under
318 anhydrous and hydrous conditions, respectively (Fig. 4d). These data are comparable with d values
319 calculated through the batch method, and similar trends are also found for L_d from CSD analysis
320 (Table 3). The morphological evolution of titanomagnetite cannot be accurately quantified through
321 the S_v^P parameter due to the extremely low crystal content and volumetric density of crystals,
322 providing standard deviations close to the measured S_v^P values. The CSD curves derived for the
323 experimental titanomagnetite grains are showed in Fig. 6 where the natural logarithm of the
324 population density is plotted against the crystal size. As a general rule, larger crystal bins develop
325 with increasing dwell time. The plots exhibit also a gentle downturn at small crystal sizes that,
326 sometimes, is accompanied by a weak upward convexity, more pronounced for crystal populations
327 from 24 h experiments. CSDs from anhydrous time-series are characterized by shallower slopes
328 than those from hydrous experiments. The anhydrous CSD curves display comparable shapes, with
329 lower slopes and intercepts at longer dwell times (Table 3). The hydrous CSD curves are also very
330 similar in terms of crystal populations and shapes, with a subtler decrease in slope and intercept
331 when the dwell time increases from 0.5 to 8 h (Table 3). At 24 h, the crystal population skews
332 significantly towards larger crystal size bins, thus defining a much smoother shape of the CSD (Fig.
333 6 and Table 3). This abrupt deviation is also related to the textural change of clinopyroxene
334 observed at the same dwell time (see discussion below).

336 3.1.2. Crystal growth kinetics

337 The value of G_{max} measured for clinopyroxene ranges from 4.98×10^{-8} to 1.49×10^{-6} cm/s
 338 and from 6.94×10^{-8} to 1.80×10^{-6} cm/s under anhydrous and hydrous conditions, respectively
 339 (Table 2). The $\log G_{max}$ vs. $\log t$ diagram shows that the growth rate decreases with increasing dwell
 340 time (Fig. 7). Data from anhydrous and hydrous isothermal time-series overlap within their
 341 uncertainties (Table 2 and Fig. 7), with the only exception observed for 24 h charges where the
 342 hydrous G_{max} is sensibly higher than the anhydrous counterpart. The linear regression fit of the data
 343 yields the following equations:

$$344 \text{ anhydrous } \log G_{max} = -0.894 \log t - 2.906 \quad (R^2 = 0.99) \quad (11)$$

$$345 \text{ hydrous } \log G_{max} = -0.851 \log t - 3.029 \quad (R^2 = 0.99) \quad (12)$$

346 The regressed data show an excellent linearity between $\log G_{max}$ and $\log t$ (Fig. 7), with a high
 347 correlation coefficient ($R^2 = 0.99$). The regression fits are characterized by positive intercepts and
 348 negative slopes that, however, do not perfectly overlap due to the different G_{max} values measured at
 349 24 h. The slope of the log-log linear trend has been proposed to give insights into the rate-limiting
 350 process for the crystal growth (i.e., interface- vs. diffusion-limited crystallization regimes;
 351 Burkhard, 2005; Orlando et al., 2008), whereas the intercept represents the logarithm of growth rate
 352 extrapolated at the time zero.

353 The growth rates of titanomagnetite crystals have been determined through the three
 354 methods described above and listed in Table 3. As the dwell time increases, G_{max} (Method 1)
 355 decreases from 1.48×10^{-7} to 6.62×10^{-9} cm/s and from 2.00×10^{-7} to 1.34×10^{-8} cm/s under
 356 anhydrous and hydrous conditions, respectively. The batch growth rate (Method 2) provides values
 357 that are one order of magnitude lower than G_{max} and ranging from 3.52×10^{-8} to 1.44×10^{-9}
 358 (anhydrous data) and from 4.64×10^{-8} to 4.04×10^{-9} (hydrous data). Similar estimates have been
 359 also derived through the CSD analysis (Method 3). As for the case of clinopyroxene,
 360 titanomagnetite crystals with higher growth rates formed over shorter dwell times (Table 3). The
 361 $\log G_{max}$ vs. $\log t$ diagram exhibits similar linear trends for both anhydrous and hydrous time-series
 362 experiments (Fig. 8) that are described by the following equations:

$$363 \text{ anhydrous } \log G_{max} = -0.787 \log t - 4.261 \quad (R^2 = 0.99) \quad (13)$$

$$364 \text{ anhydrous } \log G_{batch} = -0.843 \log t - 4.683 \quad (R^2 = 0.99) \quad (14)$$

$$365 \text{ anhydrous } \log G_{CSD} = -0.783 \log t - 4.927 \quad (R^2 = 0.98) \quad (15)$$

$$\text{hydrous } \log G_{max} = -0.731 \log t - 4.360 (R^2 = 0.98) \quad (16)$$

$$\text{hydrous } \log G_{batch} = -0.687 \log t - 5.190 (R^2 = 0.91) \quad (17)$$

$$\text{hydrous } \log G_{CSD} = -0.610 \log t - 5.519 (R^2 = 0.91) \quad (18)$$

The regression fits of growth rates calculated through Method 2 and Method 3 show comparable intercepts and slopes, attesting the reliability of estimates. Hydrous experiments show slight slope variations as a function of the different growth rates estimated for the 24 h experiment (Fig. 8).

3.2. Mineral and melt chemistry

Clinopyroxenes from anhydrous (Wo₄₃₋₄₆-En₃₇₋₄₀-Fs₁₆₋₁₉) and hydrous (Wo₄₄₋₄₇-En₃₇₋₄₁-Fs₁₅₋₁₇) experiments are classified as diopside-augite (Table 1S) according to the classification scheme of Morimoto (1988). With respect to the anhydrous charges, the compositions of crystals obtained under hydrous conditions show higher values of Si, Mg-number [Mg# = 100 × Mg/(Mg + Fe_{tot}) on molar basis] and diopside + hedenbergite (DiHd), and lower values of tetrahedrally-coordinated aluminum (^TAl), Fe³⁺, enstatite + ferrosilite (EnFs) and Ca-Tschermak + CaTi-Tschermak + CaFe-Tschermak (CaTs + CaTiTs + CaFeTs = Ts) (Fig. 9). The mineral chemistry also changes as a function of dwell time. ^TAl and Fe³⁺ are preferentially incorporated in clinopyroxenes from 0.5-2 h experiments, whereas Si content increases in crystals from 4 to 24 h charges (Fig. 9). No clear trends are observed for Ti but, from a statistical point of view, its concentration slightly decreases with dwell time, as documented for ^TAl and Fe³⁺. Overall, clinopyroxene crystals are progressively enriched in DiHd and depleted in Ts from 4 to 24 h charges (Fig. 9). The compositional change is determined by the kinetically-controlled cation exchange (Si + Mg) → (^TAl + Fe³⁺), expanding the Ts stability at the expense of Di over short time durations. Additionally, for each single experiment, strong crystal zoning is recognized either for large euhedral clinopyroxenes or dendritic crystals with analyzable cores and rims. Fig. 10a shows Di-poor and Di-rich clinopyroxene portions in which mineral compositional changes have been evidenced using a high contrast in backscattered electron images highlighting Fe-Mg crystal zoning. Fig. 10b displays also an example of electron microprobe profile analyzed across the crystal and the surrounding glass. A traverse of 5-μm-step has been performed for SiO₂, MgO, Al₂O₃, and FeO. By assuming SiO₂ and MgO variations as a proxy for Di component in clinopyroxene (Fig. 10b), the crystal zoning is characterized by Di-poor

404 (Di₄₃₋₄₄) and Di-rich (Di₄₆₋₄₈) compositions. It is apparent that complex zoning patterns develop by
405 the overgrowth of Di-rich and well-faced crystals onto early-formed Di-poor dendrites (Fig. 10a-b).

406 Titanomagnetite exhibits an almost constant composition in all the experimental charges
407 (Table 2S), showing Al₂O₃ (8.9-10.1 wt.%) and MgO (5.4-5.8 wt.%) enrichments typically
408 observed under rapid crystal growth conditions (Mollo et al., 2013b). On the other hand, the amount
409 of TiO₂ (3.8-4.2 wt.%) is relatively low, translating to an ulvospinel (Usp) content almost close to 1
410 mol.% (Table 2S). Overall, the Ti–Al–Mg cationic substitutions in titanomagnetite indicate kinetic
411 effects where Al + Mg are more easily incorporated into disequilibrium crystals by lowering the
412 number of Ti cations (Mollo et al., 2013b).

413 The residual glass composition (Table 3S) shifts from trachyandesite to basaltic
414 trachyandesite (54.1-56.1 wt.% SiO₂, 6.4-8.1 wt.% Na₂O+K₂O and Mg_{#50-54}) and from trachybasalt
415 to tephrite (45.6-48.9 wt.% SiO₂, 6.2-7.8 wt.% Na₂O+K₂O and Mg_{#50-54}) under anhydrous and
416 hydrous crystallization conditions, respectively (data on anhydrous basis according to Le Bas et al.,
417 1986). No clear glass compositional trends are found as a function of dwell time in both anhydrous
418 and hydrous charges. The interstitial glass develops within a complex crystal framework causing
419 marked compositional heterogeneity. This applies frequently in short dwell time experiments,
420 where small glass pockets are isolated by dense dendrite crystallization where the variable contents
421 and compositions of crystals induce, locally, strong glass chemical changes. On the other hand,
422 microprobe chemical profiles performed in more abundant glass portions surrounding large and
423 isolated crystals do not show concentration gradients next to the clinopyroxene surface (i.e., up to
424 30 μm far from the crystal-glass interface; Fig. 10a-b).

4. Discussion

4.1. Textural maturation of clinopyroxene

429 Under both anhydrous and hydrous conditions, the experimental charges obtained at 0.5, 2,
430 4, and 8 h are characterized by small clinopyroxenes with dendritic habits, whereas large and
431 euhedral crystals develop at 24 h. Due to the strong kinetic effects at the onset of crystallization, the
432 maximum size (65 μm on average) of clinopyroxene does not substantially change as the dwell time
433 increases from 0.5 to 8 h, whereas longer crystal lengths (100 μm on average) are measured at 24 h
434 (Table 2). Within the time interval of 0.5-8 h, the size, content and growth rate of clinopyroxene do
435 not differ significantly nor systematically between hydrous and anhydrous time-series experiments,
436 suggesting the lack of complete equilibrium. Following the theory of nucleation proposed by the
437 pioneer study of Kirkpatrick (1983), the activation energy required for the onset of nucleation is

438 proportional to the number of tetrahedral units expressed as either Al–O bonds or the most energetic
439 Si–O bonds in the crystallizing silicate melt. A great number of tetrahedral units and corner
440 connections implies a high nucleation barrier, which in turn determines high incubation times
441 (Kirkpatrick, 1983). The clinopyroxene crystal structure comprises a few tetrahedral units, so that a
442 relatively low number of structural re-arrangement is expected in the melt, translating to a short
443 nucleation delay and the attainment of high area fractions (Fig. 4a). The occurrence of dendrites is
444 addressed to melt supersaturation (e.g. Lofgren, 1974; Sunagawa, 1981) due to imposition of a
445 thermodynamic driving force promoting early crystal nucleation rather than growth (e.g. Hammer,
446 2006, 2008). Figs. 2 and 3 highlight that, upon the effect of a large degree of initial undercooling,
447 clinopyroxene dendrites from 0.5-8 h experiments are composed of tiny crystallites aligning along
448 branches that are perpendicular to each other (i.e., different orders of branch generations; Faure et
449 al., 2003). When the experimental charge is maintained isothermally for 24 h, dendrite branches
450 change into the prismatic and well-faced morphologies. The annealing treatment of 24 h before
451 quench causes melt relaxation (i.e., equilibration) phenomena that are more effective with respect to
452 the shorter dwell times (Dingwell & Webb, 1990; McMillan and Wolf, 1995; Moynihan, 1995;
453 Webb and Dingwell, 1995). Coherently, the growth of dendrites is less favored in more relaxed melt
454 regions characterized by lower degrees of melt supersaturation (i.e., “Berg effect”; Berg, 1938). The
455 lack of dendritic shapes in 24 h experiments (Fig. 3) points to a textural maturation of
456 clinopyroxene documented typically under equilibrium conditions when crystallization takes place
457 from a fully relaxed melt (Baker, 2008; Mollo et al., 2011, 2013c) or, alternatively, under extremely
458 low degrees of undercooling when the growth rate largely exceeds the nucleation rate (Simakin et
459 al., 2003; Orlando et al., 2008). Rationally, after the early effect of undercooling, the diffusion-
460 limited growth of dendrites occurring at shorter dwell times (i.e., more effective melt saturation),
461 translates to a steady-state regime under which euhedral crystals more favorably develop by
462 interface-limited growth at longer dwell times (i.e., more effective melt relaxation; Figs. 2 and 3).

463 Well-faced morphologies in natural magmatic products, as those attained in porphyritic
464 intrusions, are classically addressed to a high-temperature coarsening process of the small crystal
465 populations over time (e.g., Park and Hanson, 1999; Higgins, 1998, 1999, 2011). In crystallization
466 experiments conducted on analogue materials, Means and Park (1994) observed the evolution from
467 dendritic to blocky crystal habits as related to segmentation of the dendrite branches and further
468 coarsening of the resulting tiny crystals. A similar effect could be potentially applied to the
469 decreasing area fraction of clinopyroxene observed for anhydrous experiments (Fig. 4a). However,
470 due to the presence of scattered data affected by large uncertainties (up to $\pm 12.4\%$ on average; see
471 Table 2), no clear trends can be delineated for both anhydrous and hydrous time-series experiments

472 (Fig. 4a). Alternatively, crystal growth mechanisms by dendrite attachment have been documented
473 by both ex-situ (Pupier et al., 2008) and in-situ (Schiavi et al., 2009) laboratory observations
474 conducted on basaltic melts. More specifically, these experiments reveal that the growth histories of
475 individual crystals may proceed by intervals of relatively uniform free growth, abrupt size increase
476 by grain coalescence, and growth reduction by impingement (Schiavi et al., 2009). Coalescence
477 takes place when adjacent tiny crystals of similar orientation attach to form one
478 (crystallographically) single crystal (Pupier et al., 2008). Additionally, coalescence phenomena can
479 take place between crystals that are initially misaligned, thus involving further grain rotation
480 (Schiavi et al., 2009). As the crystal fraction increases substantially, crystal impingement and
481 overgrowth become the most important textural maturation mechanisms (Schiavi et al., 2009).
482 Further experimental studies have demonstrated that attachment of tiny dendritic crystals can
483 effectively lead to the formation of large homogeneous clinopyroxenes and plagioclases (Iezzi et
484 al., 2011, 2014; Vetere et al., 2013, 2015). This is in agreement with the theory of aggregation by
485 self-orientation of sub-micrometric crystals along crystallographic directions, in order to attain
486 minimization of interfacial energies (Kostov and Kostov, 1999; Alivisatos, 2000; Banfield et al.,
487 2000; Deer et al., 2001; Niederberger and Colfen, 2006; Sear, 2012; Teng, 2013). A similar
488 attachment feature has been observed for the formation of large plagioclase crystals due to volatile
489 exsolution and degassing of a fluid-saturated magma rising from depth (Crabtree and Lange, 2011;
490 Frey and Lange, 2011). During flowage onto the surface, naturally cooling Hawaiian and Etnean
491 lava flows may be characterized by massive crystallization (up to ~60 vol.%) of microphenocrysts
492 and microlites. In this context, the crystal growth is driven by attachments of clinopyroxene and
493 plagioclase phenocrysts with lengths up to 1 mm, having also important implications for the
494 rheological behavior of lava flows (Crisp and Baloga, 1990; Crisp et al., 1994; Cashman et al.,
495 1999; Soule et al., 2004; Lanzafame et al., 2013; Mollo et al., 2015). In naturally undercooled
496 olivines, zoning patterns reveal a crystallization history marked by early diffusion-limited growth of
497 interconnected branches forming a skeletal framework (Welsh et al., 2014; Shea et al., 2015). When
498 undercooling decreases after the fast growth stage, the skeletal branches are partially infilled to
499 yield well-faceted polyhedral crystals, as growth becomes interface-limited (Shea et al., 2015).
500 Turning to the textural maturation of clinopyroxenes from this study (Figs. 2 and 3), it cannot be
501 excluded that infilling of the early dendrite frameworks attaching to form a larger crystal takes
502 place with the annealing time. Dendritic clinopyroxene crystals may represent an early diffusion-
503 limited morphology caused by the large degree of initial undercooling and partly preserved by short
504 annealing treatments. The melt is readily supersaturated in the slow-diffusing Al cations favoring
505 the early growth of Di-poor dendrites (Fig. 10a-b) by the kinetically-controlled exchange (Si + Mg)

506 \rightarrow ($T\text{Al} + \text{Fe}^{3+}$) (Mollo et al., 2010; 2012a; 2013a). Subsequently, Di-rich overgrowths develop
507 from the residual melt that becomes substantially enriched in SiO_2 and MgO components (Fig.
508 10b). Melt relaxation and crystal attachment start to be effective phenomena when the temperature
509 is maintained isothermally and, at this condition, the interface-controlled regime becomes more
510 effective. As a consequence, early-formed Di-poor skeletal branches are partially infilled by the
511 SiO_2 - MgO -rich melt, leading to the formation of Di-rich overgrowths (Fig. 10a-b). The transition
512 between diffusion-limited to interface-limited morphologies through attachment and infilling of
513 early dendrite branches may also explain the lack of a clear trend between clinopyroxene area
514 fraction and dwell time (Fig. 4a), especially in hydrous experiments where the cation diffusivity in
515 the melt is enhanced (Zhang et al., 2010).

516 The S_v^P parameter of clinopyroxene progressively decreases with increasing dwell time
517 (Table 2 and Fig. 5), evidencing that more euhedral morphologies are attained at 24 h by the
518 textural maturation of the dendritic crystal habits. This corroborates the observation of Hammer
519 (2006, 2009) that S_v^P can be effectively used to quantify the degree of crystal euhedrality. From an
520 energetic point of view, S_v^P is directly proportional to the free energy change during melt
521 solidification and represents part of the crystallization free energy term related to the development
522 of crystal-melt interfaces (Hammer, 2006). Hence, the decreasing trend showed in Fig. 5 is an
523 expression of the minimization of the free energy as the grain boundaries approach to equilibrium
524 with the surrounding melt. This is consistent with what observed in the $\log G$ vs. $\log t$ diagrams
525 derived for clinopyroxene (Fig. 7) and titanomagnetite (Fig. 8). The slopes of the log-log linear
526 trends exhibit almost comparable values of ~ 0.7 - 0.9 that are invariably higher than the theoretical
527 threshold of 0.5 derived under diffusion-limited crystal growth conditions (Müller-Krumbhaar,
528 1975; Burkhard, 2002, 2005; Orlando et al., 2008). On this basis, it can be inferred that the decrease
529 of crystal growth with dwell time is mostly related to the progressive approach to (near-)equilibrium
530 crystallization where attachment/detachment reactions of cations from the melt onto the crystalline
531 surface (and *vice versa*) occur at the same rate. A similar interpretation has been given by Orlando
532 et al. (2008), documenting that growth kinetics of plagioclase from trachybasaltic melts are
533 generally faster than clinopyroxene, as the former exhibits steeper log-log linear trends with slopes
534 close to 1. According to Orlando et al. (2008), at 1170 °C, plagioclase equilibrates with the melt in
535 a shorter time interval of 3 h, whereas clinopyroxene requires a longer time of 20 h that,
536 importantly, is similar to the longest dwell time used in this study.

537 The values of G_{max} reported in Table 2 are highly comparable with those measured by
538 previous authors for clinopyroxenes growing from similar basaltic and trachybasaltic melts. For
539 hydrous hawaiitic melts, Simakin et al. (2003) measured G_{max} from 1.85×10^{-5} to 2×10^{-6} cm/s

540 when cooling rate experiments were quenched after 0.5 h. Such data are in agreement with the high
541 growth rates determined at the shortest dwell time adopted for the time-series experiments.
542 Conversely, the low growth rates attained at 24 h match with those derived in the fast cooling rate
543 experiments of [Burkhard \(2002\)](#) conducted on Hawaiian basalts. The in-situ observations performed
544 by [Ni et al. \(2014\)](#) confirm that, after ~1.5 h, the growth rate of clinopyroxene is comprised
545 between 1.7×10^{-6} and 6×10^{-7} cm/s, resembling values from 0.5-8 h time-series experiments.
546 Remarkably, data from [Baker \(2008\)](#) are very close to results from this study, as the author adopted
547 a comparable undercooling condition ($\Delta T = 75$ °C) and a similar trachybasaltic melt from Mt. Etna
548 volcano. For clinopyroxenes with dendritic, skeletal and swallowtail textures, [Baker \(2008\)](#)
549 estimated growth rates comprised between 5.4×10^{-7} and 9.4×10^{-8} cm/s for dwell times between
550 3.5 and 14 h. In comparison with natural studies, CSD analyses performed by [Oze and Winter](#)
551 [\(2005\)](#) on natural basalts yield clinopyroxene growth rates variable from 3.74×10^{-6} to 1.00×10^{-8}
552 cm/s over crystal residence times of ~0.04-9 h ([Table 2](#)). In contrast, the static experiments of
553 [Burkhard \(2005\)](#) on re-heated anhydrous basaltic glasses (dwell times of 22-576 h at 930-990 °C
554 and 1 atm), and the low undercooling ($\Delta T = 20$ °C) experiments of [Orlando et al \(2008\)](#) on seeded
555 anhydrous trachybasalts (dwell times of 3-40 h at 1170 °C and 1 atm), resulted in growth rates up to
556 one order of magnitude lower than those from this study ([Fig. 11](#)). According to a great number of
557 authors (e.g. [Dowty, 1980](#); [Lofgren & Russell, 1986](#); [Leshner et al., 1999](#); [Conte et al., 2006](#); [Mollo](#)
558 [et al., 2013b](#)), rapid crystallization kinetics are usually attained in experiments employing high
559 degrees of undercooling or cooling rate. The lower slopes of log-log trends from [Orlando et al.](#)
560 [\(2008\)](#) stands for the smaller degrees of undercooling used to approach equilibrium crystal sizes
561 with respect to dendrites developed during time-series experiments ([Fig. 11](#)). This reflects distinct
562 crystallization mechanisms operating at different degrees of undercooling (e.g. [Kirkpatrick, 1975](#);
563 [Sunagawa, 2005](#); [Faure et al., 2007](#); [Ni et al., 2014](#)). Indeed, at low degrees of undercooling, the
564 textural maturation proceeds through screw dislocation and layer-by-layer growth determining
565 smooth growth surfaces and euhedral morphologies, while at higher degrees of undercooling crystal
566 nucleation dominates resulting in dendritic and skeletal morphologies. Log-log trends from the
567 experiments of [Burkhard \(2005\)](#) show an overall slope of ~1 ([Fig. 11](#)), confirming the achievement
568 of equilibrium crystal sizes for clinopyroxenes annealed over 20-24 h.

570 4.2. Textural maturation of titanomagnetite

571 Titanomagnetite crystals have strong tendency to attain euhedral habits at the shortest dwell
572 time of 0.5 h ([Table 3](#)). The extremely fast textural maturation documented in this study can be
573 addressed to the high interfacial energies occurring between oxide crystals and silicate melts

574 (Hammer, 2006). This effect is enhanced by the application of an isothermal temperature after rapid
575 cooling rate conditions, as demonstrated by Mollo et al. (2012a) who compared two sets of identical
576 cooling rate experiments, the first immediately quenched after cooling and the second kept at the
577 final temperature over a certain time. It is also important to note that small titanomagnetites
578 nucleate, sometimes, on the tip of branches formed by clinopyroxene dendrites (Fig. 3). According
579 to cooling rate experiments conducted on basaltic melts by Hammer et al. (2010), there is a
580 preferential epitaxial relationship between the $\{110\}_{\text{tmt}}$ and $\{010\}_{\text{cpx}}$ faces, resulting from
581 heterogeneous nucleation of tiny titanomagnetite crystals on dendrite branches of clinopyroxene.
582 Moreover, compositional gradients arising during dendrite growth may provide a further
583 thermodynamic driving force for heterogeneous nucleation of chemically different phases
584 (Kirkpatrick, 1975; Walker et al., 1976; Hammer, 2006; Mollo et al., 2012b). This textural
585 relationship becomes less frequent with increasing dwell time, especially when the textural
586 maturation produces larger and euhedral clinopyroxene crystals (Fig. 3). Energetically unfavorable
587 grain boundaries develop as the growth rate prevails over the nucleation rate, causing an increase in
588 the surface energy associated with the crystal-crystal interface relative to the crystal-melt one
589 (Hammer et al., 2010). The maximum size (10 μm on average) of titanomagnetite slightly changes
590 from 0.5 to 8 h, but then increases significantly (23 μm on average) in hydrous experiments
591 equilibrated for 24 h (Table 3). The titanomagnetite growth rates ($G_{\text{max}} > G_{\text{batch}} \approx G_{\text{CSD}}$) invariably
592 decrease with increasing dwell time (Table 3 and Fig. 8). This relationship is confirmed by the
593 intercept values of log-log trends, representing growth rates extrapolated at the time zero. The value
594 of G_{max} is up to five times higher than those of G_{CSD} and G_{batch} , resembling data (from 1.2×10^{-8} to
595 3.5×10^{-9} cm/s) obtained from the heating experiments of Burkhardt (2002, 2005). With respect to
596 natural titanomagnetites studied by Oze and Winter (2005), the growth rates from CSD analysis of
597 time-series experiments largely overlap those estimated from phenocryst populations modelled at
598 residence times of 0.12-6 h.

599 Log-log linear trends for titanomagnetite show slopes shallower than those derived for
600 clinopyroxene (see Eq.12a-c and Eq.13a-c; Figs. 7 and 8), indicating prolonged growth of the
601 former crystal phase (Orlando et al., 2008). This is also evidenced by the counterclockwise rotation
602 of the CSD curves as a function of time (Fig. 6). Although the CSD analysis of experimental
603 products cannot be directly matched to that of natural rocks, the evolution of scale invariant
604 parameters (such as the slope) allows to evaluate the interplay between nucleation and growth
605 mechanisms. The fan-like evolution of CSDs is typically observed in natural and experimental
606 igneous systems (e.g. Waters & Boudreau, 1996; Marsh, 1998; Pupier et al., 2008; Schiavi et al.,
607 2009), in which an increased importance of growth vs. nucleation reflects the progressive evolution

608 of the melt composition and the decreasing liquidus temperature with increasing crystal content
609 (e.g. [Cashman, 1993](#); [Hammer, 2008](#)). Generally, for equivalent dwell times, titanomagnetite CSD
610 curves from hydrous experiments exhibit slopes slightly shallower than crystal populations from
611 anhydrous charges. Under hydrous crystallization conditions and high degrees of undercooling,
612 [Muncill and Lasaga \(1988\)](#) observed an increased crystal growth rate with respect to early
613 anhydrous experiments ([Muncill & Lasaga, 1987](#)), responding to crystallization kinetics enhanced
614 by the effect of water on cation diffusivity in the melt (e.g., [Dowty, 1980](#); [Davis et al., 1997](#);
615 [Lasaga, 1998](#); [Hammer, 2008](#); [Calzolaio et al., 2010](#); [Zhang, 2010](#)). Nevertheless, at dwell times of
616 24 h, a striking difference is observed in the titanomagnetite crystalline population, exhibiting
617 skewness towards larger sizes and drastically shallower slopes ([Fig. 6](#) and [Table 3](#)). In this respect,
618 the attainment of euhedral textures for clinopyroxene crystals at the same dwell times may have
619 influenced the textural development of titanomagnetite crystals, causing shift from heterogeneous
620 nucleation to a growth-dominated crystallization regime. Intriguingly, a downturn of the CSD curve
621 is observed in almost all the experiments at the smaller crystal size bins ([Fig. 6](#)). This effect has
622 been previously interpreted as an inadequate spatial resolution of the analyzed images ([Hammer et
623 al., 1999](#); [Higgins, 2006](#); [Armienti, 2008](#)), the evidence of textural coarsening (e.g., [Higgins, 1998,
624 1999, 2011](#); [Ni et al., 2014](#)), or a decrease of the nucleation rate with decreasing the residual melt
625 volume ([Marsh, 1998](#)). However, BSE microphotographs show a high resolution that, in most
626 cases, is less than one micron. Moreover, the observed fan-like evolution of CSD curves cannot be
627 accounted by the sole decrease of nucleation rate with time that, in turn, is mostly restricted to area
628 fractions higher than 50% in a closed system ([Marsh, 1998](#)). Indeed, the ex-situ cooling
629 experiments of [Pupier et al., \(2008\)](#) and the in-situ textural observations conducted by [Schiavi et al.
630 \(2009\)](#) attest that fanning evolution and the concave downward curve at the small grain size results
631 from a decrease in the number of small grains in favor of large grains over time (i.e., Ostwald
632 ripening and grain coalescence).

633 634 4.3. *Crystal growth vs. cation diffusivity in the melt*

635 One intriguing feature from time-series experiments presented in this study is the lack of
636 clear concentration gradients in the glass surrounding the clinopyroxene crystals ([Fig. 10a-b](#)) that is
637 apparently in contrast with the complex textures and compositions, as well as the overall increase in
638 DiHd relative to Ts with increasing dwell time ([Fig. 9](#)). It is known that a high degree of
639 undercooling favors a diffusion-limited crystallization regime accompanied by the development of a
640 diffusive boundary layer in the melt next to the growing crystal face ([Dowty, 1980](#); [Lofgren and
641 Smith, 1980](#); [Kirkpatrick, 1981](#); [Hammer, 2008](#); [Mollo and Hammer, 2017](#)). The deviation from
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642 homogeneous equilibrium increases, as the crystal growth rate greatly exceeds the diffusion rates of
643 chemical species in the melt (Kirkpatrick, 1981). Compositional gradients develop in the melt
644 feeding the crystal growth, so that cations compatible with the crystal lattice are depleted in the melt
645 near the crystal surface, whereas the incompatible cations are rejected from the advancing crystal
646 surface and enriched in the adjacent melt (Kirkpatrick, 1981). Mollo et al. (2013a, 2013c)
647 conducted cooling rate experiments on the same trachybasaltic melt used in this study, with the
648 difference that the charges were immediately quenched at the end of cooling and no dwell times or
649 annealing times were applied before quenching. The authors observed that the composition of
650 clinopyroxene is sympathetic with the enrichment or depletion of cations in the diffusive boundary
651 layer supplying nutrients to the advancing crystal surface, responding to the cation exchange ($\text{Si} +$
652 Mg) \rightarrow ($^{\text{T}}\text{Al} + \text{Fe}^{3+}$). It is worth stressing that the thickness of the diffusive boundary layer at the
653 crystal-melt interface decreases with dwell time, as the chemical species are more efficiently
654 rejected away from the crystal surface (Dowty, 1980; Lofgren and Smith, 1980). The melt becomes
655 fully relaxed and all chemical gradients cease when the diffusing chemical components have
656 sufficient time to re-equilibrate with the original bulk composition of the far-field melt (Kirkpatrick,
657 1981; Baker, 2008). Disequilibrium cation partitioning during rapid crystal growth may also takes
658 place in presence of a very thin diffusive boundary layer around the growing crystals (Watson and
659 Muller, 2009). This is especially true for low viscosity melts, such as trachybasalts at Mt. Etna
660 volcano, so that the thin diffusive boundary layer is limited to the region affected by cation
661 interactions with the crystal surface (Watson and Muller, 2009). According to the experiments of
662 Baker (2008), after initial undercooling, the clinopyroxene growth rate significantly decreases with
663 time, causing the concentration gradients in the melt adjacent to the crystal surface to decrease by
664 diffusive relaxation. The same applies to melt inclusions where their entrapment require interface-
665 limited phenomena where an early rapid crystal growth forms melt embayments. Subsequently, a
666 period of slow growth is necessary to seal and isolate the embayment during interface-limited
667 textural maturation (e.g., Stewart and Pearce 2004; Blundy and Cashman 2005). Data from this
668 study confirm that G_{max} decreases by one order of magnitude with dwell time from 0.5 to 24 hours,
669 favoring the shift from a diffusion-limited to an interface-limited crystallization regime (Fig. 7).
670 Therefore, the lack of an analyzable diffusive boundary layer next to the crystal surface may be
671 addressed to 1) the decreasing crystal growth rate with dwell time 2) the fast diffusivity of cations
672 in the trachybasaltic melt and 3) the development of a thin diffusive boundary layer close to the
673 spatial resolution limits of the microprobe. The diffusion coefficients (D) of cations compatible (Mg
674 and Ca) and incompatible (Al and Na) with clinopyroxene crystal lattice have been calculated
675 through the equations 23 ($D_{Mg} = 5-6 \times 10^{-13} \text{ m}^2/\text{s}$), 24 ($D_{Ca} = 6-7 \times 10^{-12} \text{ m}^2/\text{s}$) and 36 ($D_{Al} = 8-9 \times$

676 10^{-14} m²/s) reported in the review study of Zhang et al. (2010), as well as through equation SEBD
677 ($D_{Na} = 7-8 \times 10^{-10}$ m²/s) reported in Chen and Zhang (2009). Note that these equations were derived
678 by the regression fit of anhydrous experimental data, consequently, the calculated diffusion
679 coefficients represent a conservative limit for the cation mobility in the melt that, in turn, is
680 enhanced by the effect of H₂O. The isothermal temperature of 1,100 °C was used as input data for
681 calculations, together with the average analyses of glasses from experimental charges, representing
682 the quenched melt feeding the clinopyroxene growth. Assuming absence of convection fluxes in the
683 experimental capsules, the cation diffusion length d was derived as follows (Baker, 2008):

$$d = \sqrt{D t} \quad (19)$$

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687 In the log-log diagram presented in Fig. 12a, the diffusion length ($\log d$) is plotted against the
688 experimental dwell time ($\log t$). During clinopyroxene growth by dendrite attachment, the size of
689 single and isolated dendrites varies from ~5 to ~50 μm under both anhydrous and hydrous
690 conditions. This range is comparable with the $\log d$ calculated for Al and Mg at 0.5 and 1 h (Fig.
691 12a). In contrast, at 24 h, the dendrite size is always lower than d , suggesting that the diffusivity in
692 the melt is fast enough to supply fresh cations from the far-field (relaxed) melt to the advancing
693 crystal surface (Fig. 12a). Moreover, from 0.5 to 24 h dwell times, all the dendrites with size lower
694 than 12 μm do never intercept the travelling distance of Al (i.e., the slow-diffusing incompatible
695 cation). Evidently, when the experimental charges were cooled from the superliquidus condition
696 down to the isothermal temperature of 1,100 °C, the melt was rapidly supersaturated in Al by the
697 fast nucleation and growth of Di-poor dendrites. However, the diffusive boundary layer developing
698 at the dendrite-melt interface was extremely thin, so that the system readily shifted from diffusion-
699 limited to interface-limited conditions, favoring the overgrowth of Di-rich crystals. To test this
700 hypothesis, the 1-dimensional “disequilibrium” crystal growth equation derived by Watson and
701 Muller (2009) has been used to model how the early enrichment of Al in clinopyroxene dendrites is
702 controlled by the thickness of the diffusive boundary layer in the melt. Importantly, Al is rate-
703 limiting for the disequilibrium growth of clinopyroxene due to its slow diffusivity in the melt phase
704 (cf. Iezzi et al., 2011, 2014; Mollo et al., 2013a, 2013c). Moreover, Al is incompatible with
705 clinopyroxene crystal lattice and its concentration increases in the diffusive boundary layer as the
706 crystal growth rate increases. Data from this study show that the partition coefficient (K_{Al}) of Al
707 measured at the clinopyroxene-glass interface decreases from 0.77 to 0.45 with increasing dwell
708 time. According to the kinetically-controlled cation exchange ($\text{Si} + \text{Mg} \rightarrow \text{}^T\text{Al} + \text{Fe}^{3+}$), the
709 concentration of ${}^T\text{Al}$ in clinopyroxene decreases from ~0.43 apfu for the disequilibrium (Di-poor)

710 dendrites to ~0.21 apfu for the near-equilibrium (Di-rich) overgrowths (Fig. 9). The disequilibrium
711 incorporation of Al in rapidly growing dendrites can be modelled through the following expression
712 (Watson and Muller, 2009):

$$713 \quad 714 \quad \text{Disequilibrium } Al_2O_3 \text{ in dendrites} = \frac{\text{Equilibrium } Al_2O_3 \text{ in overgrowths}}{1 - (1 - K_{Al}) \frac{C_{max,d}}{D_{Al}}} \quad (20)$$

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716 Considering that melt supersaturation phenomena refer mostly to the fast growth rate ($1.49\text{-}1.80 \times$
717 10^{-6} cm/s; Table 2) of clinopyroxene, the thickness of the diffusive boundary layer controlling the
718 disequilibrium uptake of Al in early-formed dendrites varied from ~1 to ~2 μm as a function of K_{Al}
719 (Fig. 12b). Therefore, the boundary layer enriched in slow-diffusing incompatible cations and
720 feeding the growth of tiny Di-poor dendrites may have rapidly disappeared after a relatively short
721 time, being replaced by Di-rich overgrowths forming under near-equilibrium conditions.

722 5. Conclusive remarks

723 Textural and compositional data presented in this experimental study underline the
724 importance of crystallization kinetics in the solidification of silicate melt. Due to the early effect of
725 undercooling, the system attempts to approach a more thermodynamically convenient configuration
726 of crystal-melt interfaces. Particularly, results from time-series experiments points towards a
727 mineral textural evolution dominated by extremely fast crystal growth rates. Near-equilibrium
728 crystal contents are also attained over relatively short dwell times of a few hours. The crystallization
729 proceeds isothermally by minimizing the interfacial free energy between melt and crystal
730 morphologies characterized by high surface to the volume ratio of tiny dendrites. The implications
731 for volcanic systems are of first order relevance, especially when considering the development of
732 porphyritic textures. For example, nearly constant temperatures may enhance the textural evolution
733 of phenocrysts in magmas located at shallow-to-intermediate crustal depths where strong
734 undercooling phenomena may take place. Overall, the most important outcomes from this study can
735 be summarized as follows:

- 736 1) the textural evolution of clinopyroxene is dominated by progressive maturation of an early
737 formed Di-poor, Al-rich dendritic fabric that, however, is characterized by a slightly
738 changing area fraction (45% on average) from 0.5 to 24 h;
- 739 2) the attachment of clinopyroxene skeletal branches is accompanied by infilling and
740 overgrowth of Di-rich, Al-poor crystal portions due to the shift from a diffusion-limited to
741 an interface-limited growth regime;

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- 3) the textural maturation by infilling and overgrowth agrees with the minimization of crystal-melt interfaces suggested by the decreasing value of S_v^P with dwell time, as well as the lack of a clear increasing crystallinity;
- 4) the euhedral morphology of titanomagnetite immediately develop at the early stage of crystal nucleation and growth;
- 5) the CSD analysis of titanomagnetite crystals points to a textural evolution by coarsening;
- 6) in 0.5-8 h experiments, the early dendritic growth of clinopyroxene seems to facilitate the heterogeneous nucleation of titanomagnetite grains;
- 7) conversely, in 24 h experiments, the coarsening of titanomagnetite appears related to the achievement of a near-equilibrium morphology for the clinopyroxene.

Figure captions

Fig. 1. Explanatory illustration of the image segmentation procedure. Original BSE image showing clinopyroxene (Cpx), titanomagnetite (Timt) and glass (Gl) (a). Thresholding procedure conducted through the ImageJ code and binarization of the image identifying clinopyroxene (black), titanomagnetite (red) and glass (white). Cpx, clinopyroxene. Timt, titanomagnetite. Gl, glass.

Fig. 2. Textural development of clinopyroxene crystals from anhydrous and hydrous time-series experiments. Equal-area best-fit ellipses (with major and minor axes) are derived by the ImageJ code and superimposed on the segmented clinopyroxene crystals.

Fig. 3. Backscattered electron (BSE) images of experimental run products from anhydrous and hydrous time series. Cpx, clinopyroxene. Timt, titanomagnetite. Gl, glass

Fig. 4. Variations of clinopyroxene and titanomagnetite textural parameters with dwell time. Clinopyroxene area fraction (a) and maximum crystal size (b). Titanomagnetite area fraction (c) and maximum crystal size (d). Cpx, clinopyroxene. Timt, titanomagnetite.

Fig. 5. Variation of surface area to volume ratio (S_v^P) parameter of clinopyroxene with dwell time for both anhydrous (red diamonds) and hydrous (blue diamonds) time-series experiments.

Fig. 6. Crystal Size Distribution (CSD) analysis of titanomagnetite for both anhydrous (red diamonds) and hydrous (blue diamonds) time-series experiments.

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778 Fig. 7. $\log G_{max}$ vs. $\log t$ diagram based on clinopyroxene data from anhydrous and hydrous
779 experiments.

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782 Fig. 8. $\log G$ vs. $\log t$ diagram based on titanomagnetite data from anhydrous (a) and hydrous (b)
783 experiments.

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786 Fig. 9. Clinopyroxene compositional variation as a function of dwell time. $Mg\# = 100 \times Mg / (Mg +$
787 $Fe_{tot})$ on molar basis. DiHd = diopside (Di) + hedenbergite (Hd). EnFs = enstatite (En) + ferrosilite
788 (Fs). Ts = Ca-Tschermak (CaTs) + CaTi-Tschermak (CaTiTs) + CaFe-Tschermak (CaFeTs).

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791 Fig. 10. Clinopyroxene Di-poor and Di-rich compositional variations revealed by backscattered
792 electron (BSE) microphotographs (a). The high contrast used for the image highlights as Fe cations
793 in clinopyroxene backscatter electrons better than Mg ones. Example of electron microprobe profile
794 (i.e., 5- μ m-step traverses) for SiO₂, MgO, Al₂O₃, and FeO analyzed across the crystal and
795 surrounding glass (b). The complex zoning pattern is characterized by overgrowth of Di-rich and
796 well-faced crystals onto early-formed Di-poor dendrites.

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799 Fig. 11. $\log G$ vs. $\log t$ diagram in which clinopyroxene data from this study are compared with those
800 from the experimental studies of Orlando et al. (2008) and Burkhard (2005).

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803 Fig. 12. Log-log diagram showing the diffusion length ($\log d$) plotted against the experimental dwell
804 time ($\log t$) (a). The size of single and isolate dendrites is also reported for comparison as a grey
805 area. Modelling data from the equation of [Watson and Muller \(2009\)](#) showing the control of the
806 thickness (d) of the diffusive boundary layer in the melt on the Al₂O₃ concentration of
807 clinopyroxene (b).

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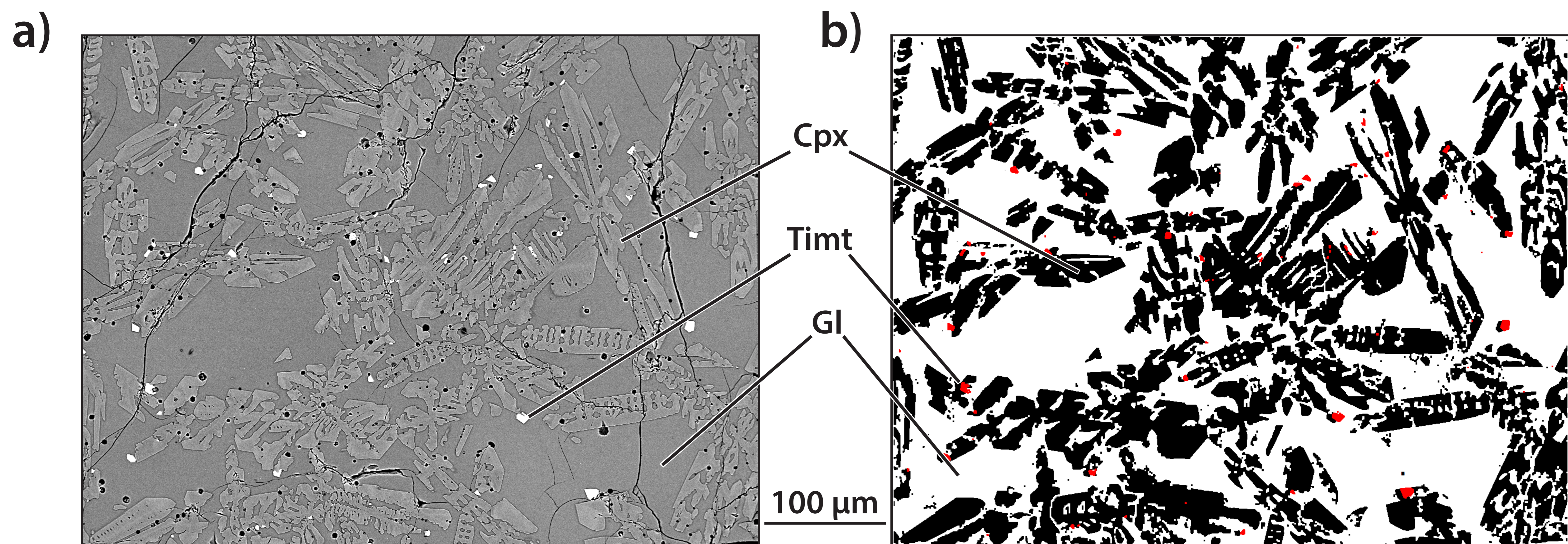


Fig. 1. Explanatory illustration of the image segmentation procedure. Original BSE image showing clinopyroxene (Cpx), titanomagnetite (Timt) and glass (Gl) (a). Thresholding procedure conducted through the ImageJ code and binarization of the image identifying clinopyroxene (black), titanomagnetite (red) and glass (white). Cpx, clinopyroxene. Timt, titanomagnetite. Gl, glass.

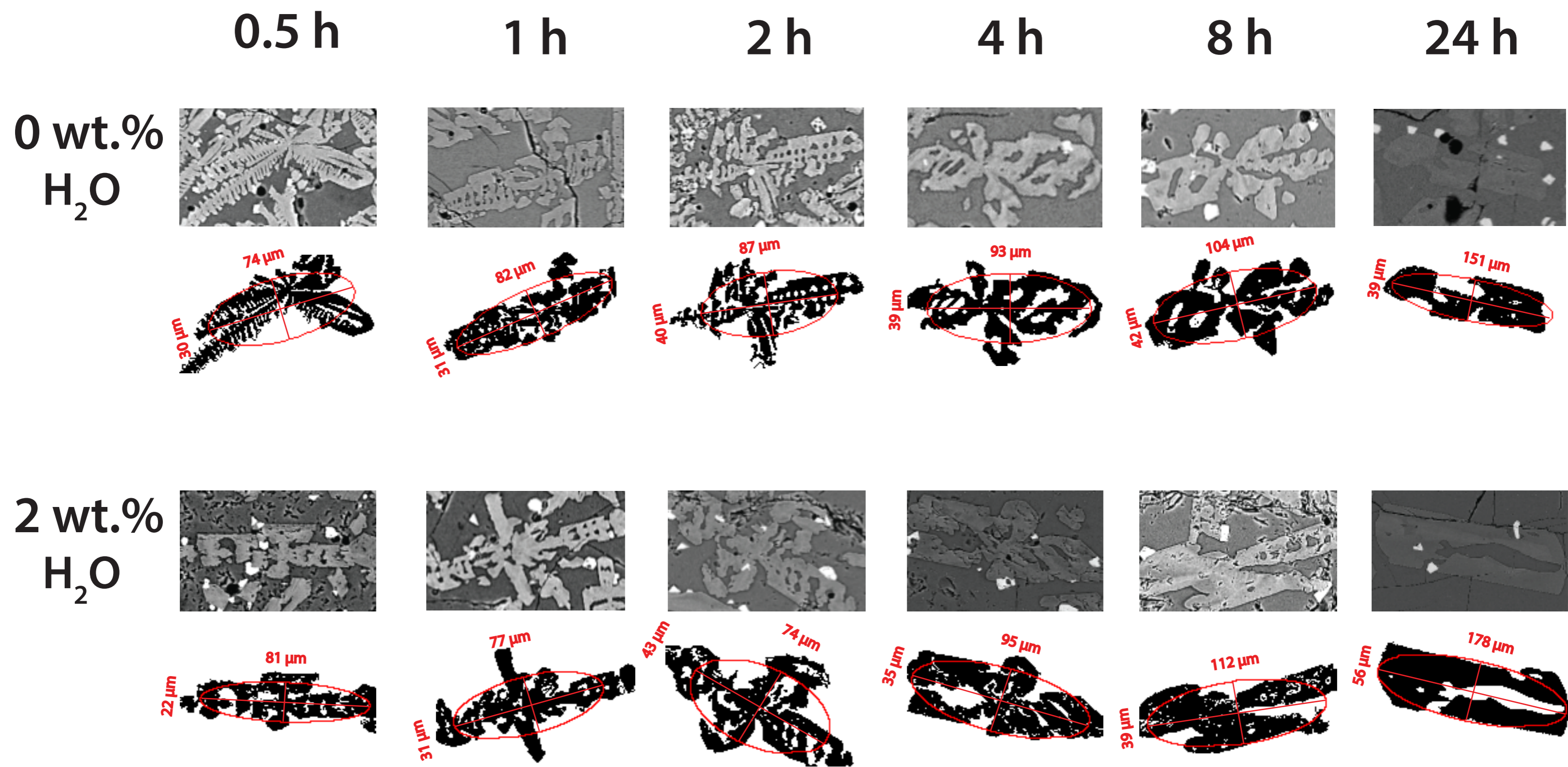


Fig. 2. Textural development of clinopyroxene crystals from anhydrous and hydrous time-series experiments. Equal-area best-fit ellipses (with major and minor axes) are derived by the ImageJ code and superimposed on the segmented clinopyroxene crystals.

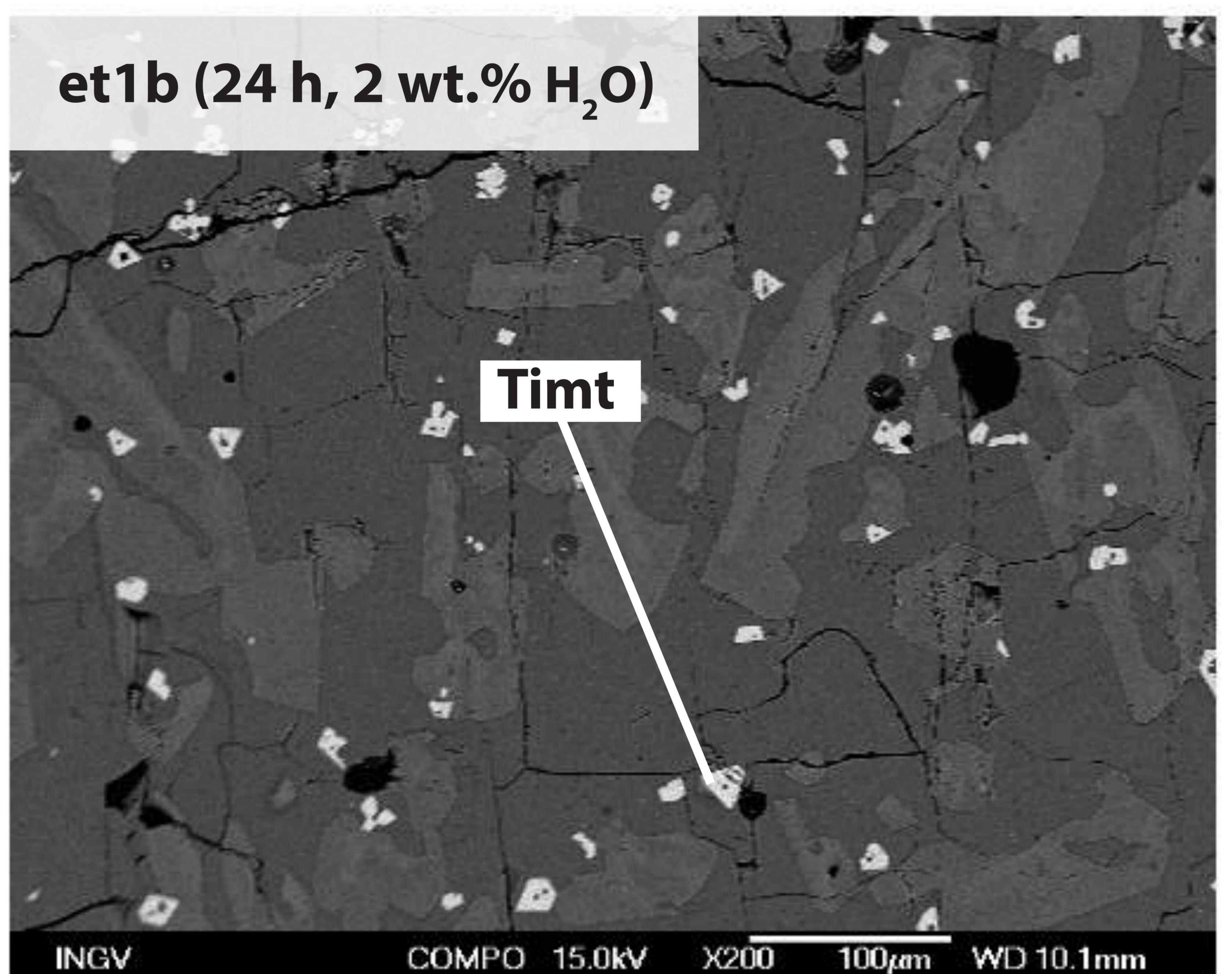
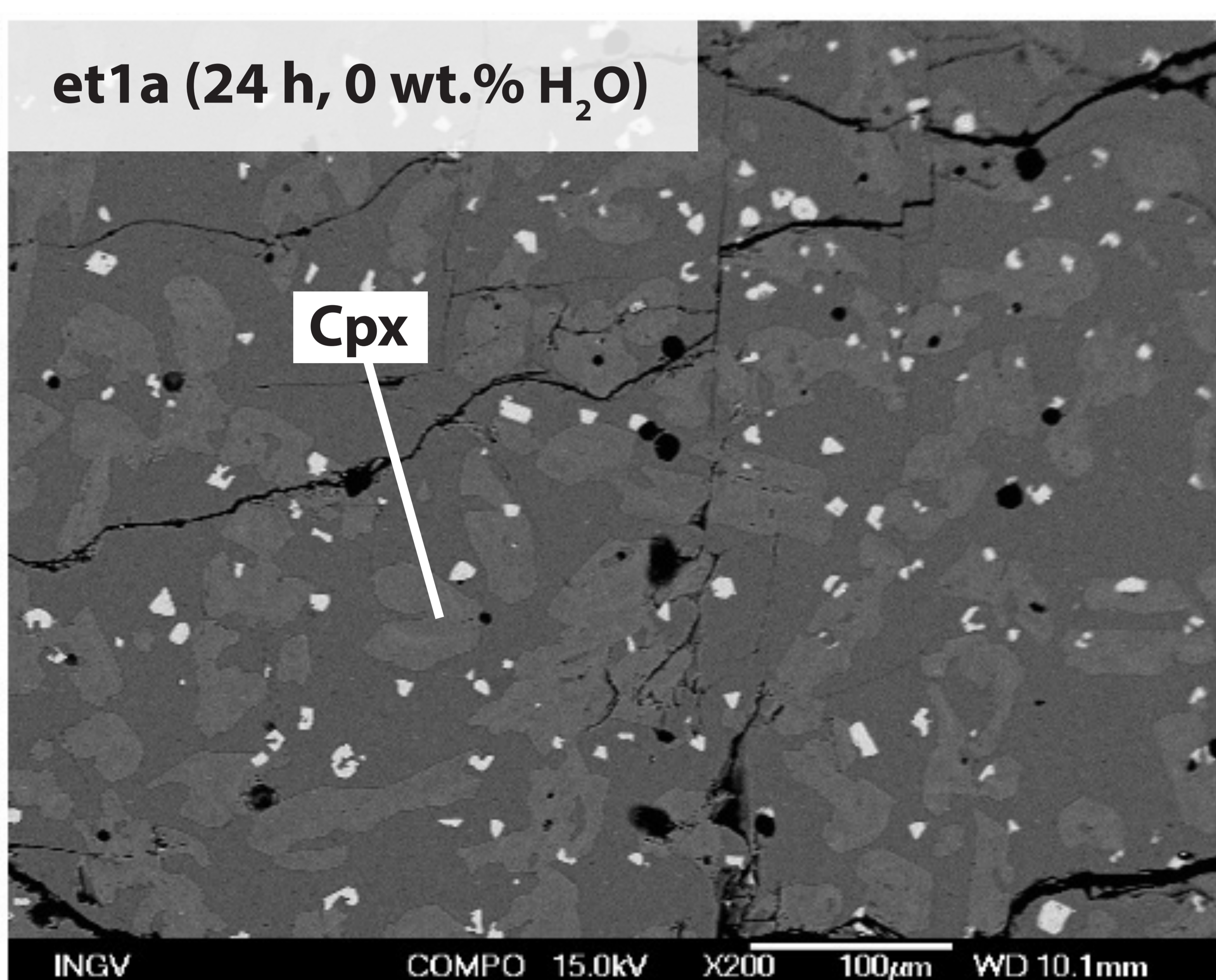
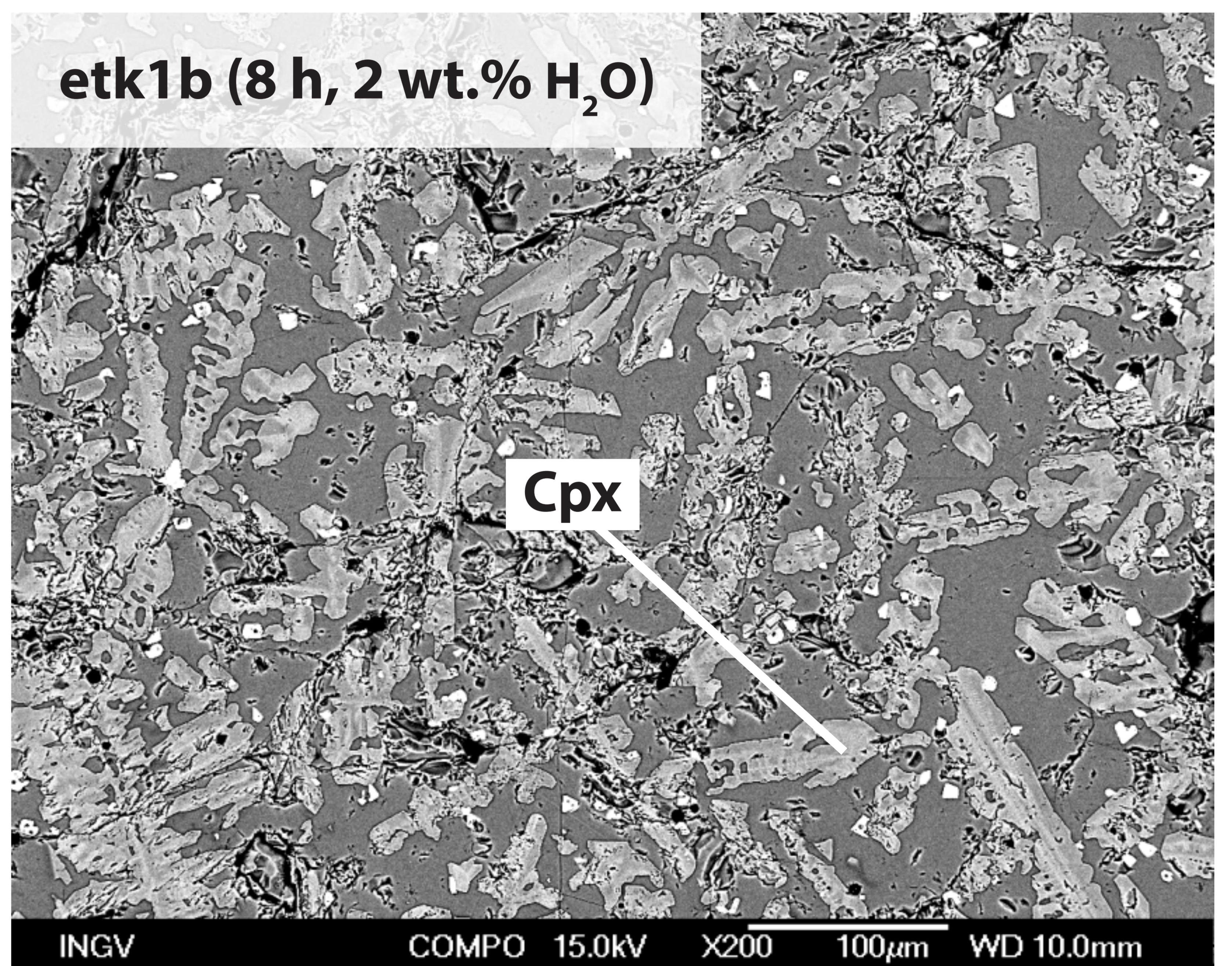
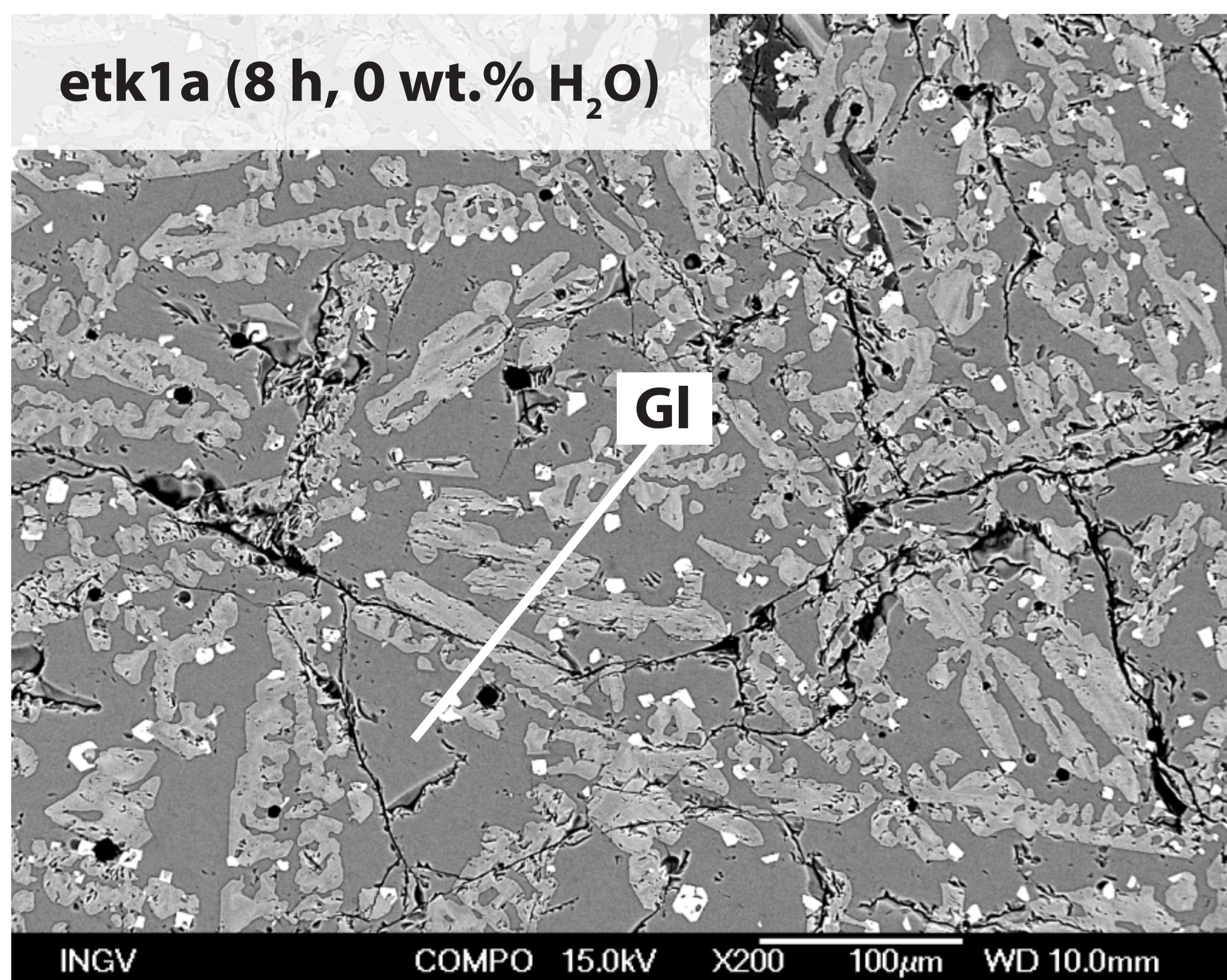
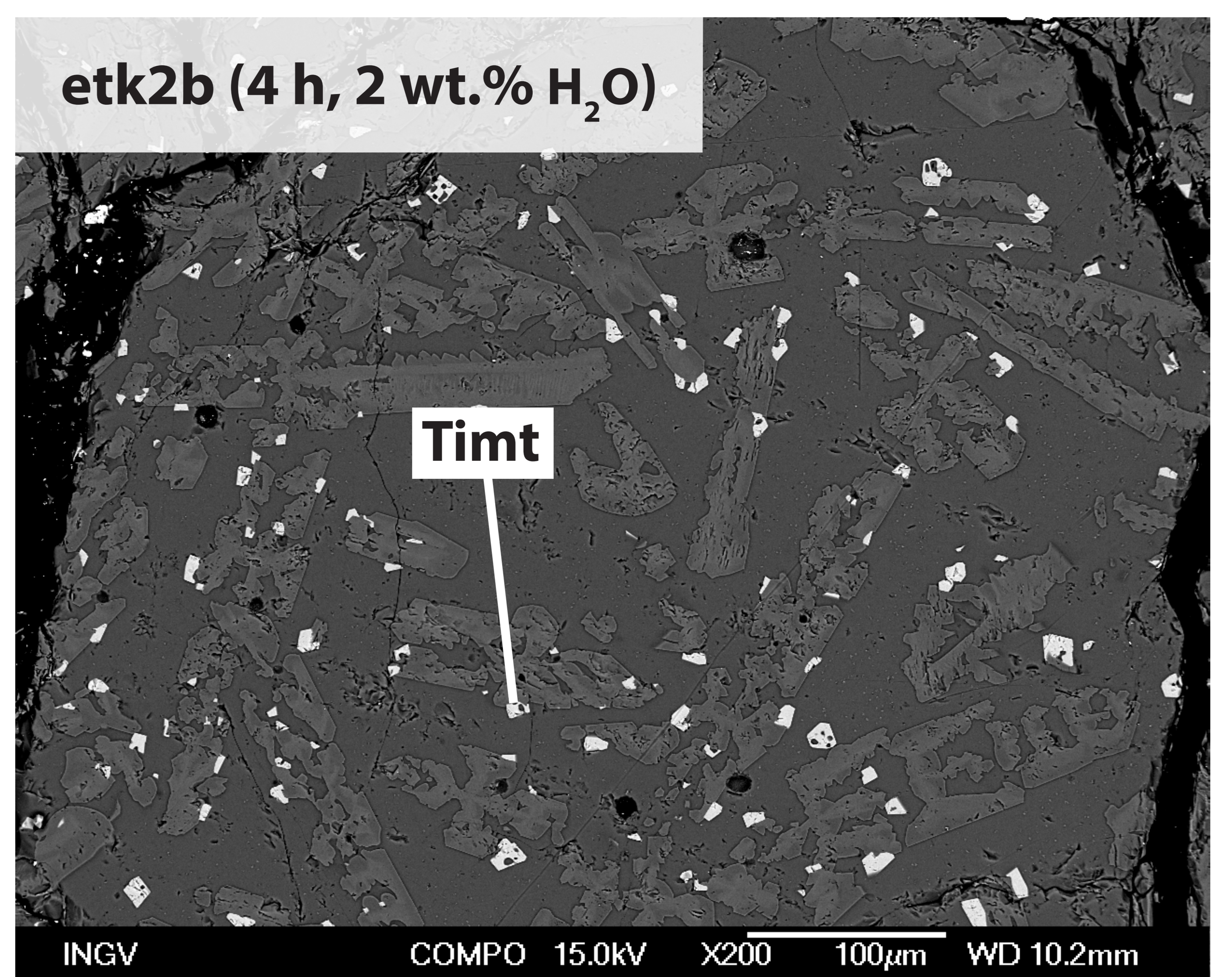
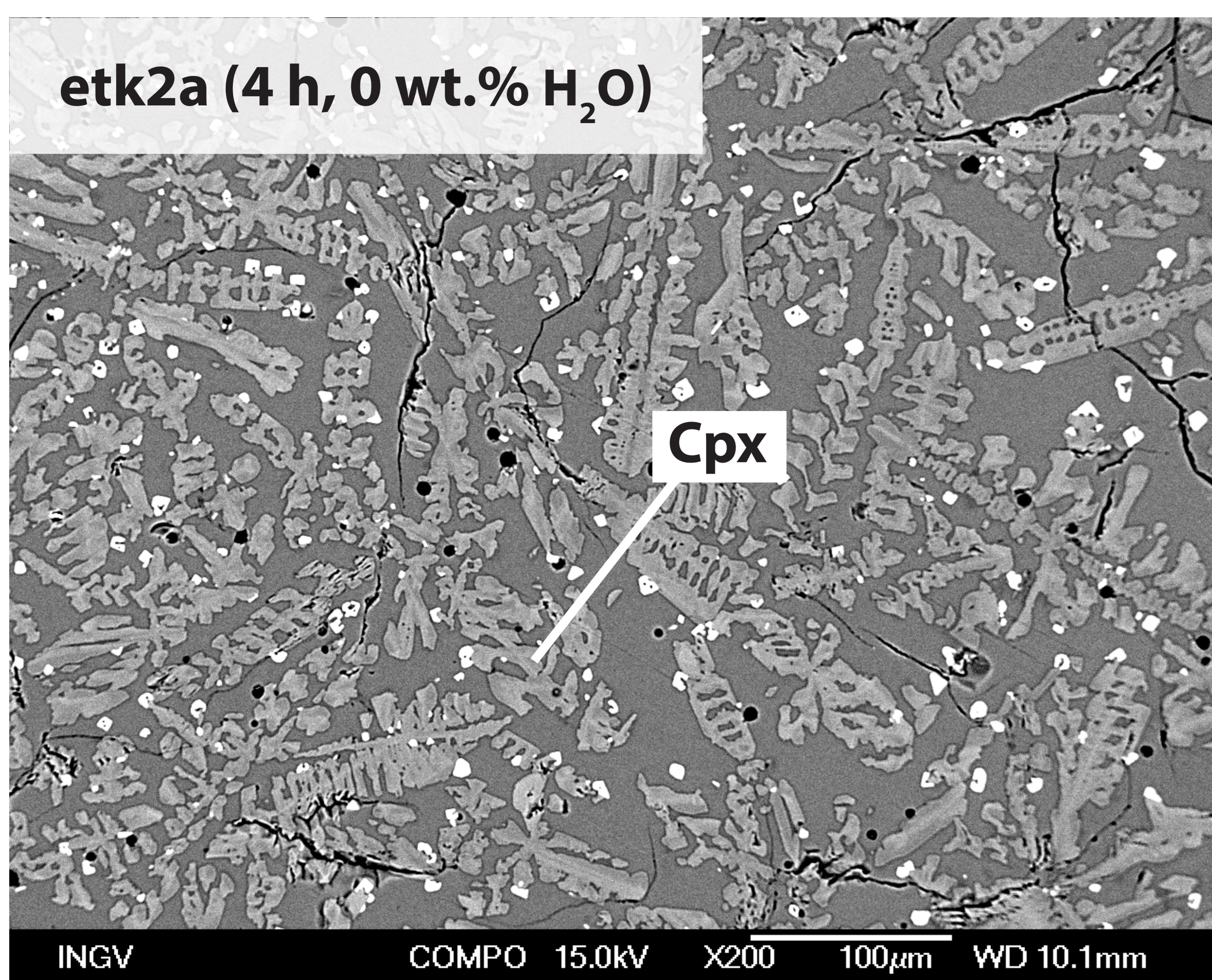


Fig. 3. Backscattered electron (BSE) images of experimental run products from anhydrous and hydrous time series. Cpx, clinopyroxene. Timt, titanomagnetite. Gl, glass

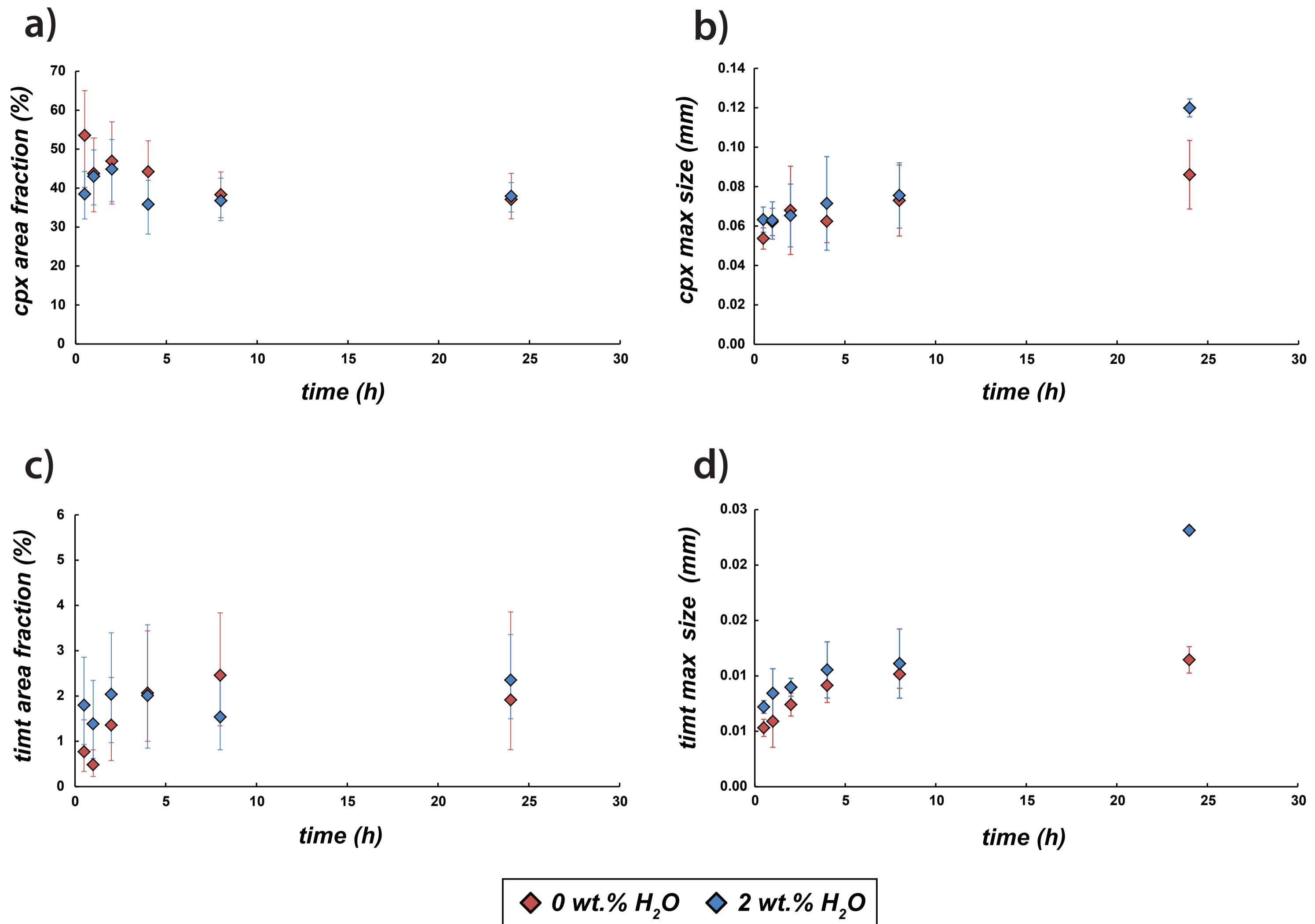


Fig. 4. Variations of clinopyroxene and titanomagnetite textural parameters with dwell time. Clinopyroxene area fraction (a) and maximum crystal size (b). Titanomagnetite area fraction (c) and maximum crystal size (d). Cpx, clinopyroxene. Timt, titanomagnetite.

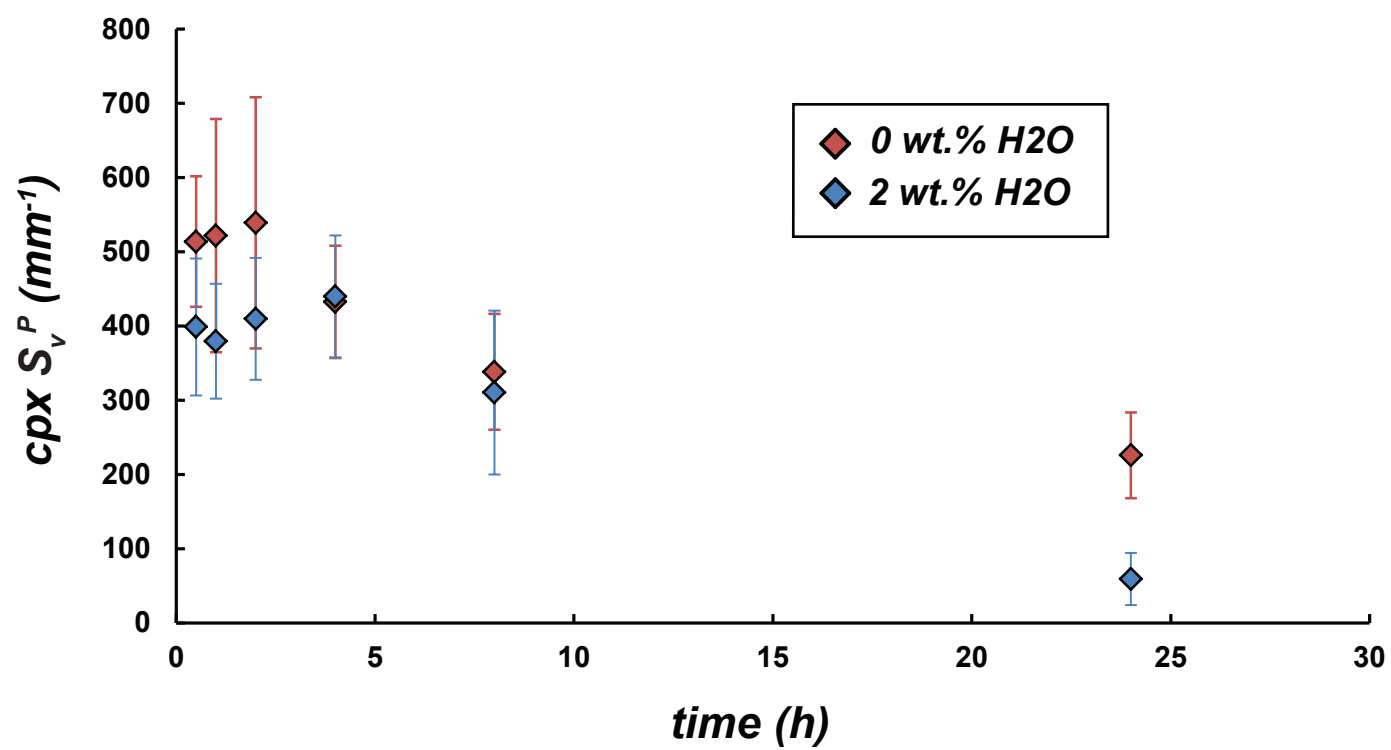


Fig. 5. Variation of surface area to volume ratio (S_v^P) parameter of clinopyroxene with dwell time for both anhydrous (red diamonds) and hydrous (blue diamonds) time-series experiments.

Figure 6 with caption - 2 columns

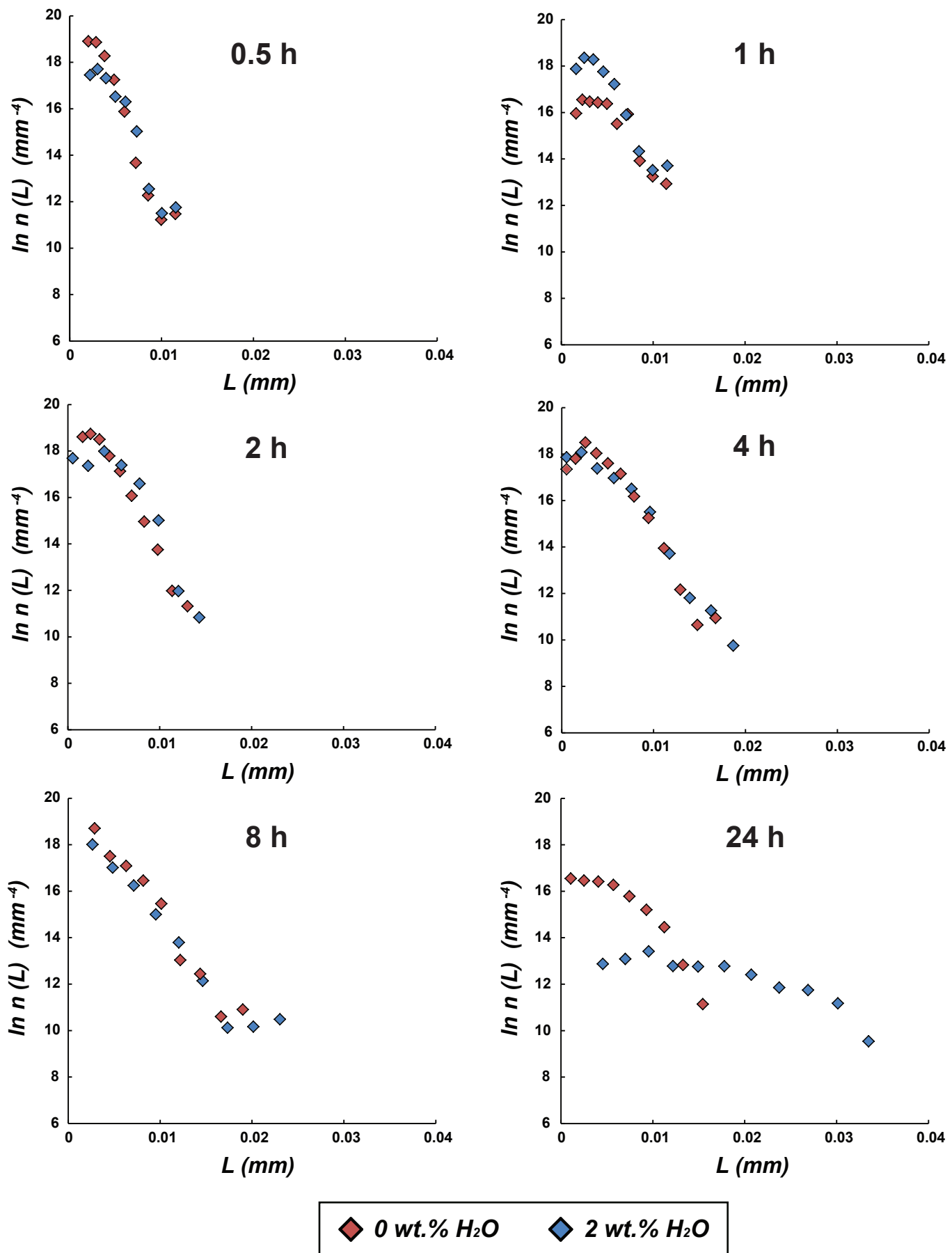


Fig.6. Crystal Size Distribution (CSD) analysis of titanomagnetite for both anhydrous (red diamonds) and hydrous (blue diamonds) time-series experiments.

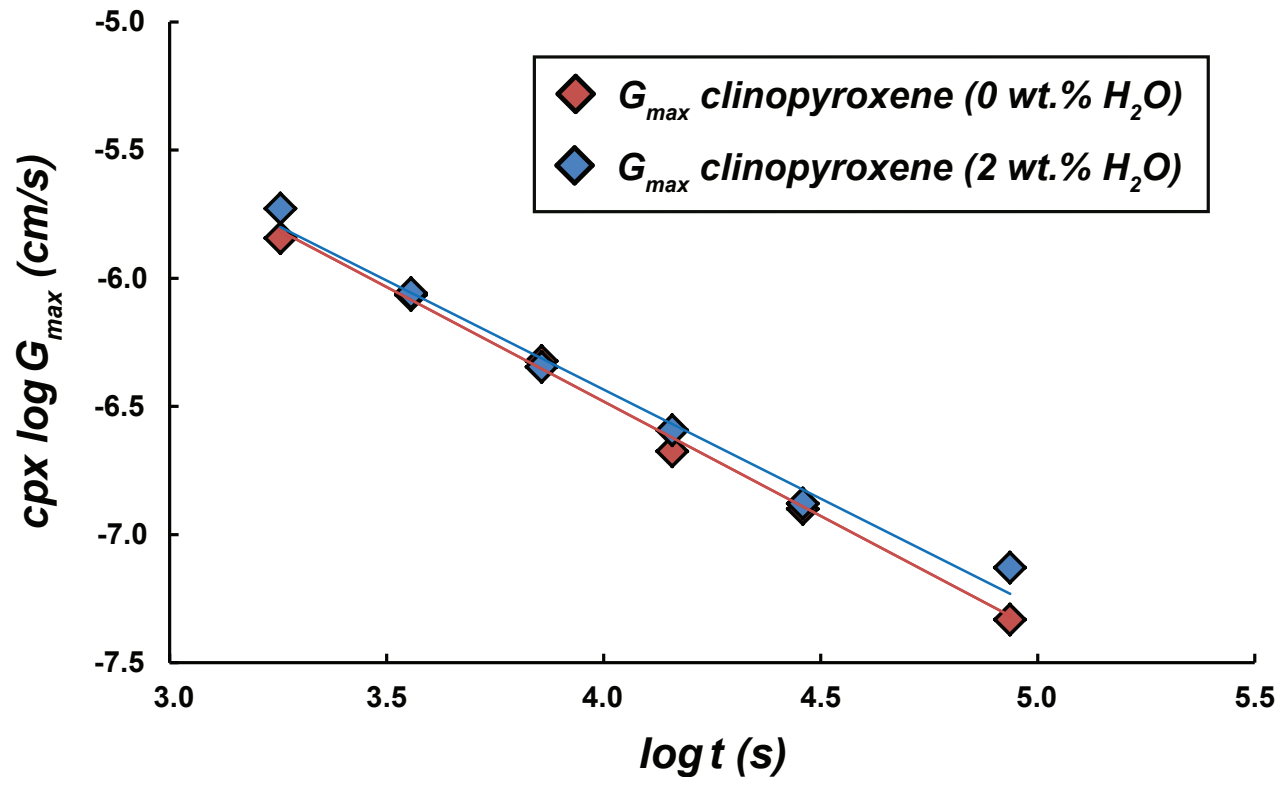


Fig. 7. $\log G_{\text{max}}$ vs. $\log t$ diagram based on clinopyroxene data from anhydrous and hydrous experiments.

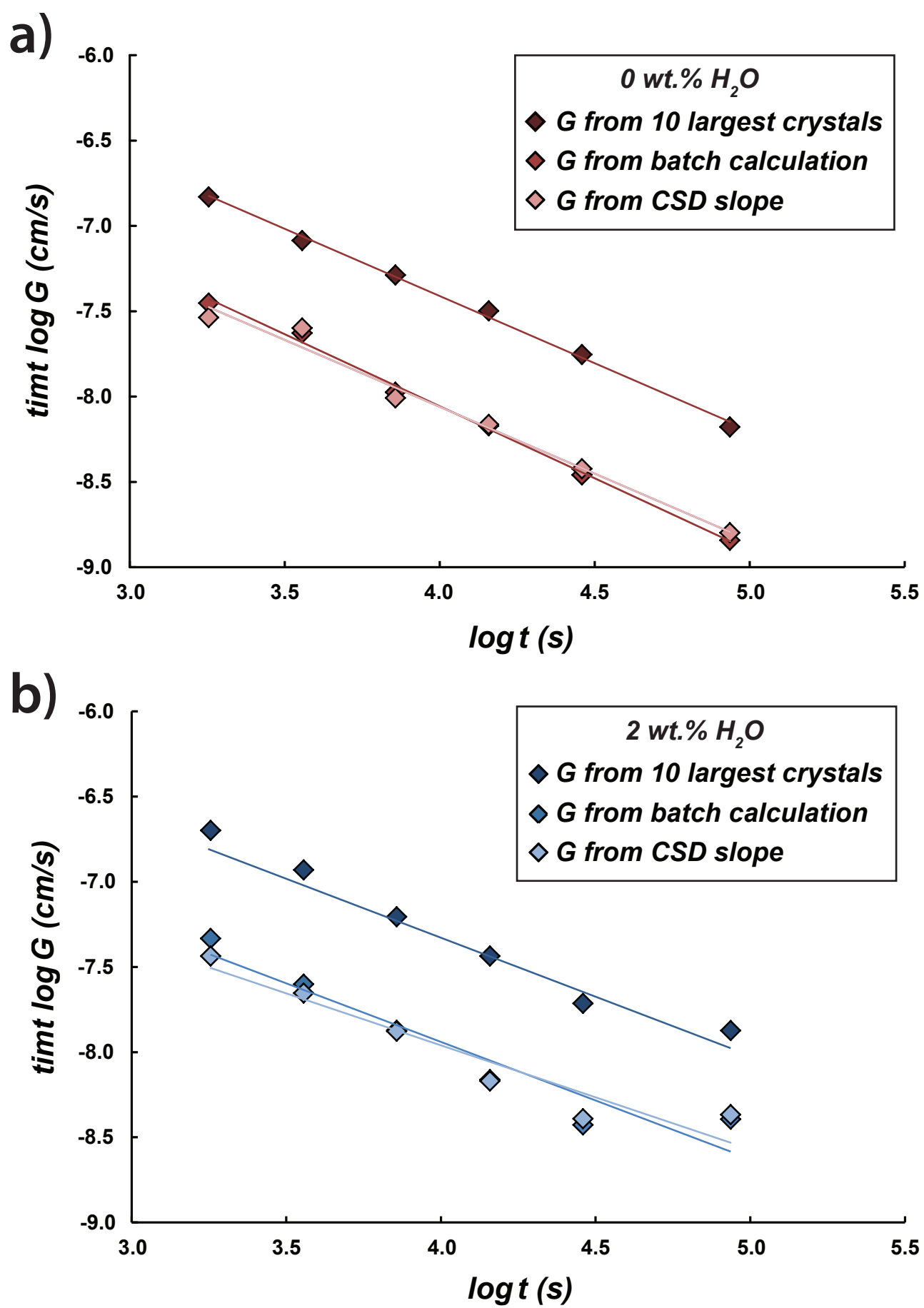


Fig. 8. $\log G$ vs. $\log t$ diagram based on titanomagnetite data from anhydrous (a) and hydrous (b) experiments.

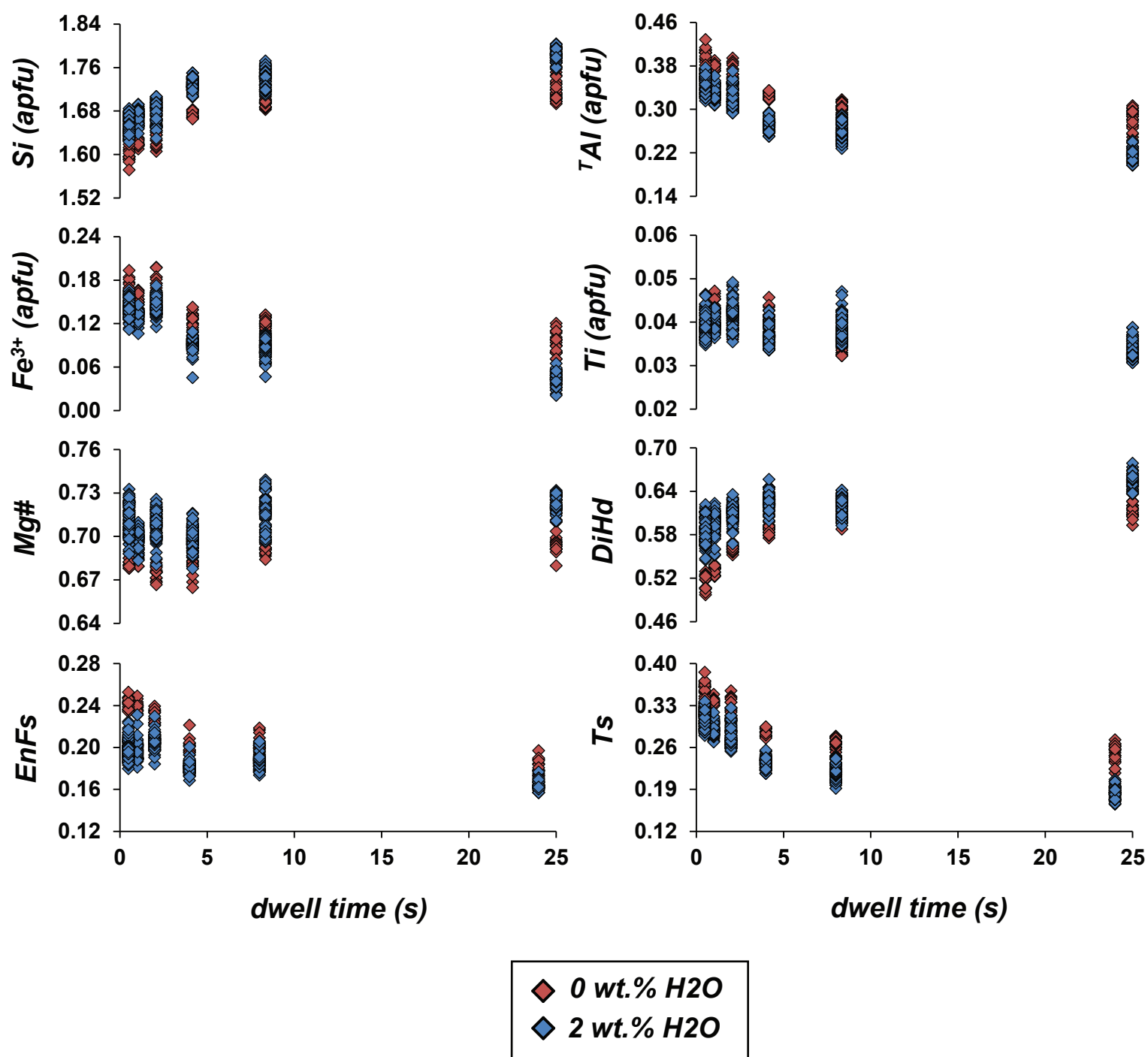


Fig. 9. Clinopyroxene compositional variation as a function of dwell time. $Mg\# = 100 \times Mg / (Mg + Fe_{tot})$ on molar basis. DiHd = diopside (Di) + hedenbergite (Hd). EnFs = enstatite (En) + ferrosilite (Fs). Ts = Ca-Tschermak (CaTs) + CaTi-Tschermak (CaTiTs) + CaFe-Tschermak (CaFeTs).

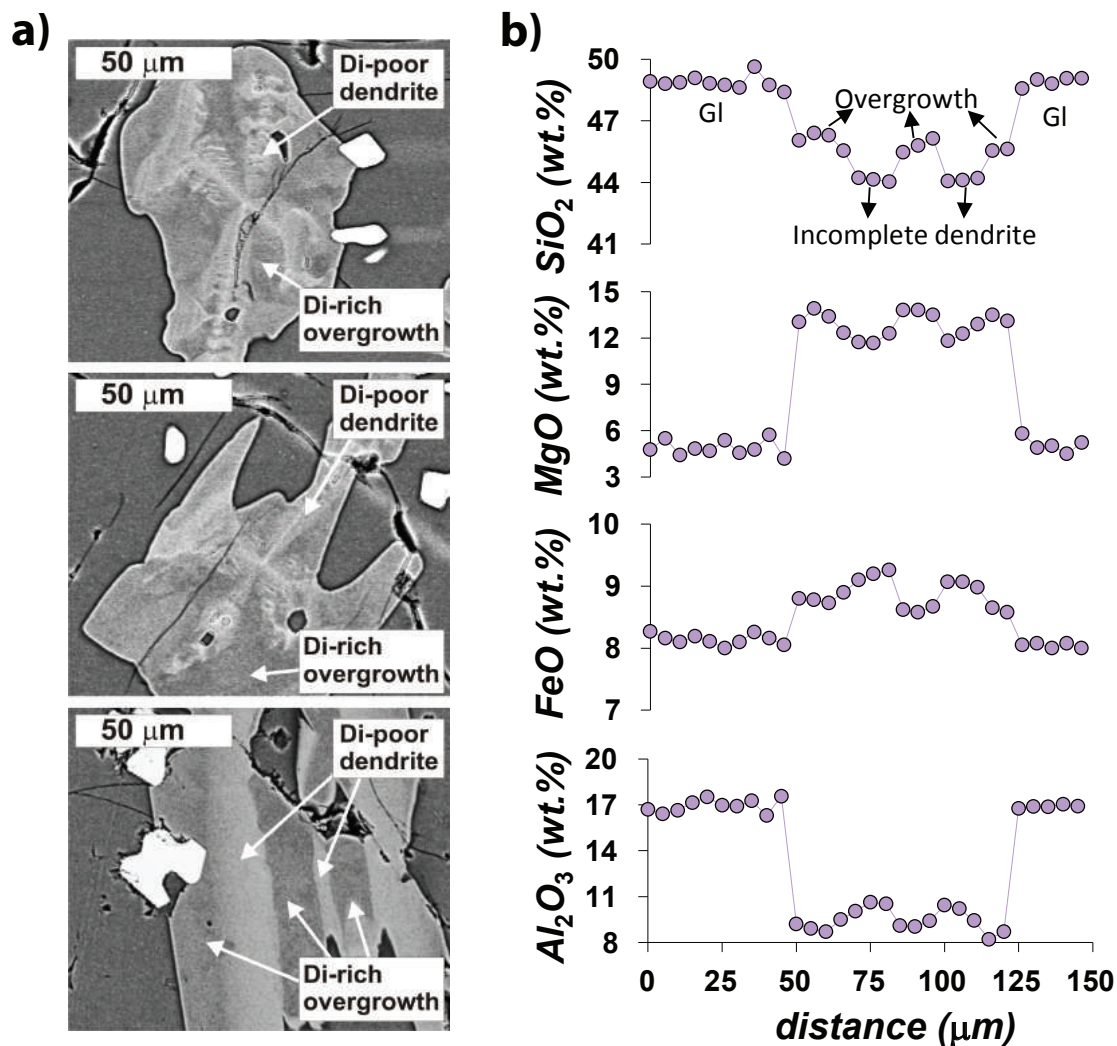


Fig. 10. Clinopyroxene Di-poor and Di-rich compositional variations revealed by backscattered electron (BSE) microphotographs (a). The high contrast used for the image highlights as Fe cations in clinopyroxene backscatter electrons better than Mg ones. Example of electron microprobe profile (i.e., 5- μm -step traverses) for SiO₂, MgO, Al₂O₃, and FeO analyzed across the crystal and surrounding glass (b). The complex zoning pattern is characterized by overgrowth of Di-rich and well-faced crystals onto early-formed Di-poor dendrites.

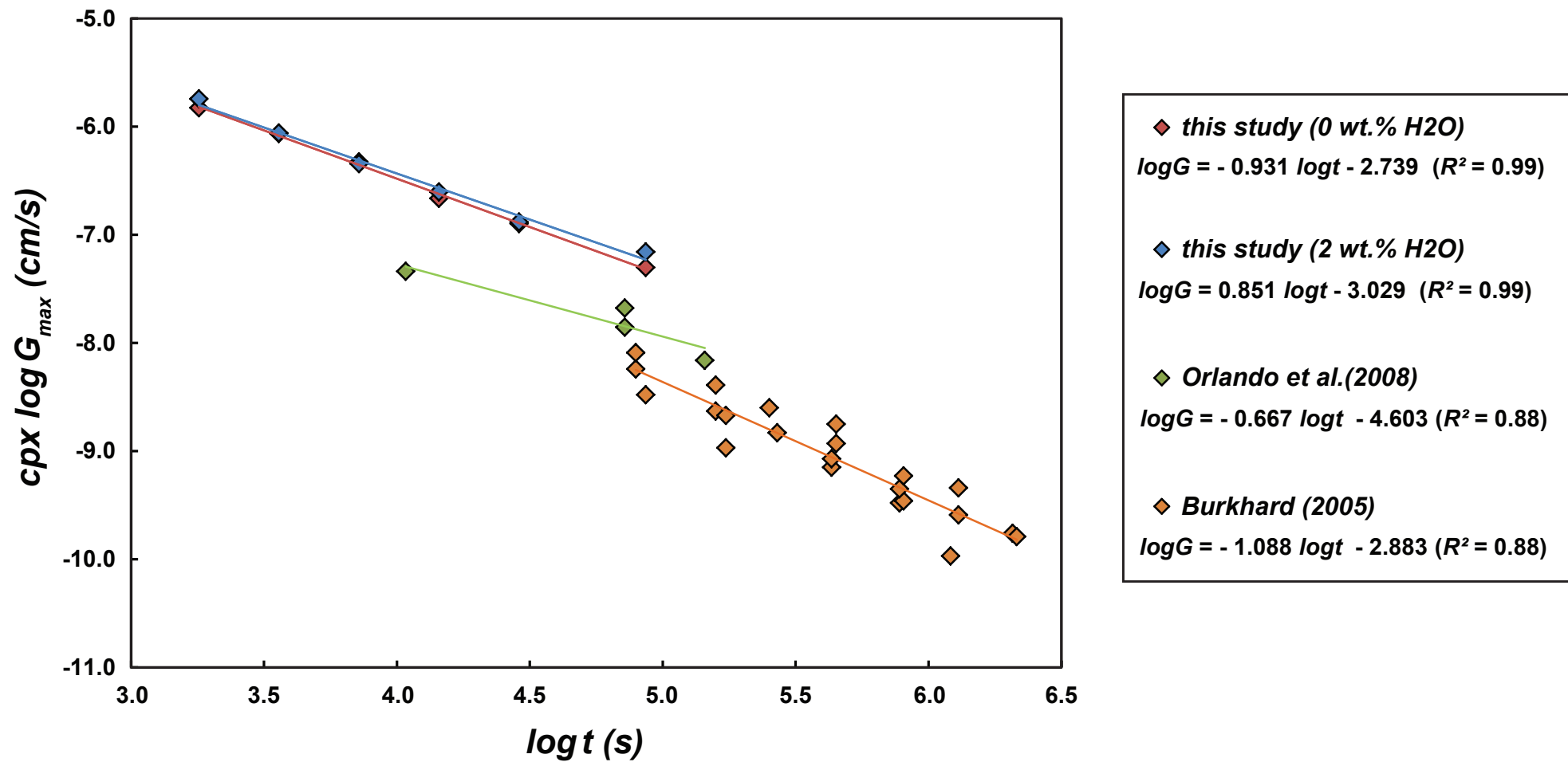


Fig. 11. $\log G$ vs. $\log t$ diagram in which clinopyroxene data from this study are compared with those from the experimental studies of Orlando et al. (2008) and Burkhard (2005).

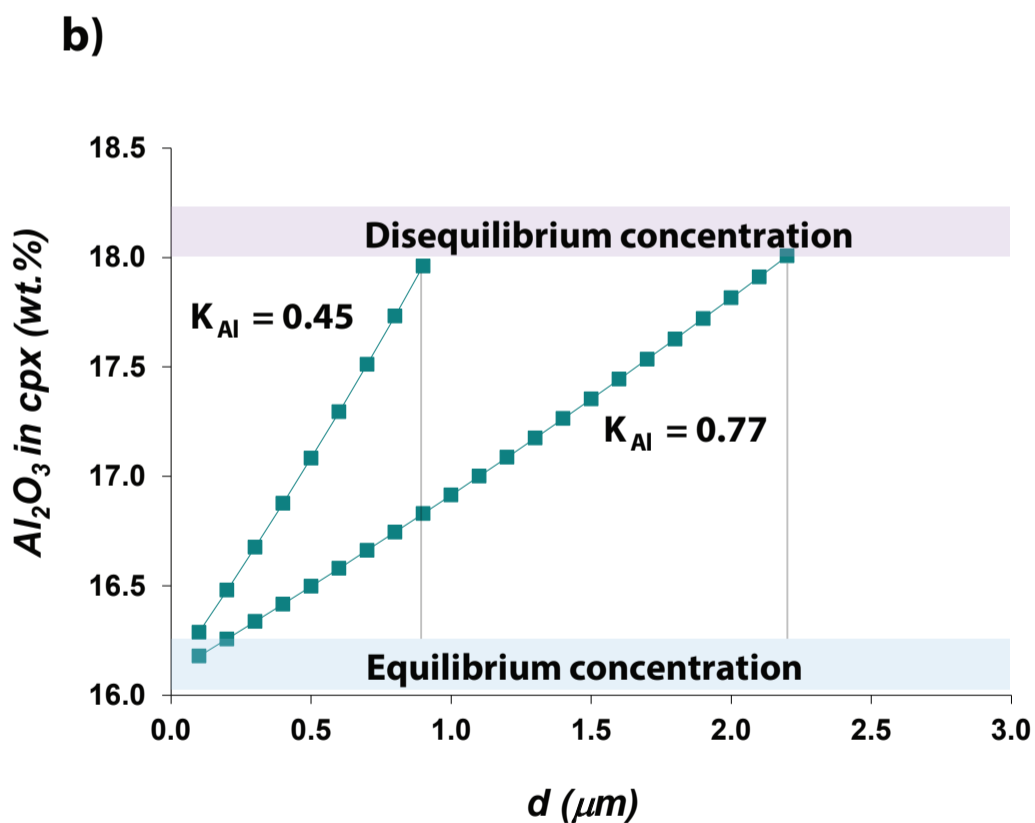
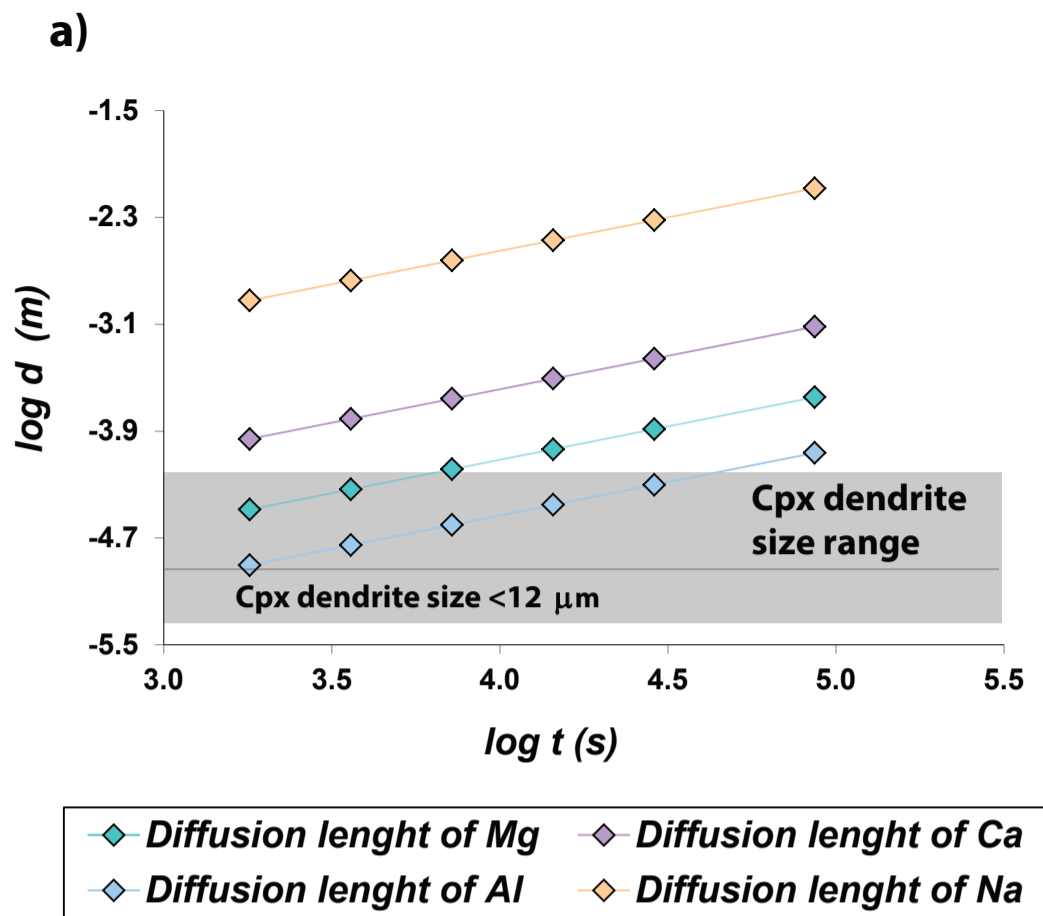


Fig. 12. Log-log diagram showing the diffusion length ($\log d$) plotted against the experimental dwell time ($\log t$) (a). The size of single and isolate dendrites is also reported for comparison as a grey area. Modelling data from the equation of Watson and Muller (2009) showing the control of the thickness (d) of the diffusive boundary layer in the melt on the Al_2O_3 concentration of clinopyroxene (b).

Table 1[Click here to download Table: Table 1_Pontesilli et al.docx](#)

Run (#)	P (MPa)	T (°C)	H ₂ O (wt.%)	time (h)	<i>f</i> O ₂ redox buffer
etk5a	400	1100	0	0.5	NNO+2
etk4a	400	1100	0	1	NNO+2
etk3a	400	1100	0	2	NNO+2
etk2a	400	1100	0	4	NNO+2
etk1a	400	1100	0	8	NNO+2
et1a	400	1100	0	24	NNO+2
etk5b	400	1100	2	0.5	NNO+2
etk4b	400	1100	2	1	NNO+2
etk3b	400	1100	2	2	NNO+2
etk2b	400	1100	2	4	NNO+2
etk1b	400	1100	2	8	NNO+2
et1b	400	1100	2	24	NNO+2

Table 1. Run conditions adopted for the time-series experiments.

Table 2[Click here to download Table: Table 2_Pontesilli et al.docx](#)

Run (#)	H ₂ O (wt.%)	time (h)	area fraction		L _{max}		G _{max} × 10 ⁷		S _v ^P (Hammer, 2006)	
			(%)	σ	(μm)	σ (10)	(cm/s)	σ (10)	(mm ⁻¹)	σ
etk5a	0	0.5	53.53	12.41	53.66	5.39	14.91	1.50	513.84	87.93
etk4a	0	1	43.73	9.48	62.07	6.96	8.62	0.97	521.89	157.36
etk3a	0	2	46.92	10.57	67.96	22.39	4.72	0.16	539.32	169.19
etk2a	0	4	44.20	8.33	62.40	10.83	2.17	0.38	432.59	75.62
etk1a	0	8	38.31	5.89	72.92	18.03	1.27	0.30	338.42	78.16
et1a	0	24	37.10	5.83	86.01	17.39	0.50	0.10	225.98	57.83
etk5b	2	0.5	38.46	6.09	63.24	6.41	18.02	2.49	398.81	92.31
etk4b	2	1	43.03	7.06	62.82	9.49	8.73	1.32	379.58	77.33
etk3b	2	2	44.87	7.96	65.34	15.96	4.54	1.11	409.97	82.13
etk2b	2	4	35.82	6.90	71.45	23.73	2.48	0.82	439.87	82.16
etk1b	2	8	36.76	5.44	75.52	16.57	1.31	0.29	310.44	110.32
et1b	2	24	37.92	3.76	119.93	4.60	0.69	0.03	59.33	34.98

Table 2. Textural parameters and growth rates estimated for clinopyroxene. L_{max}: maximum crystal length, average of 10 largest crystals. G_{max}: maximum crystal growth rate, average of 10 largest crystals. S_v^P: surface area to volume ratio, calculated according to Hammer (2006). See text for equations defining the parameters.

Table 3[Click here to download Table: Table 3_Pontesilli et al.docx](#)

Run	H ₂ O	time	area fraction		L _{max}	G _{max} × 10 ⁷		G _{batch} × 10 ⁷	G _{CSD} × 10 ⁷	d	L _d	N _A	CSD		
(#)	(wt.%)	(h)	(%)	σ	(μm)	σ (10)	(cm/s)	σ (10)	(cm/s)	(cm/s)	d (μm)	L _d (μm)	(mm ⁻²)	slope (mm ⁻¹)	intercept (mm ⁻⁴)
etk5a	0	0.5	0.77	0.68	5.31	0.78	1.48	0.22	0.35	0.29	2.54	1.04	1174.07	-957.42	21.67
etk4a	0	1	0.48	0.41	5.90	2.35	0.82	0.28	0.24	0.25	3.39	1.81	205.46	-551.91	18.86
etk3a	0	2	1.36	1.21	7.41	1.04	0.51	0.07	0.11	0.10	3.05	1.41	1519.44	-708.21	20.64
etk2a	0	4	2.07	1.81	9.15	1.55	0.32	0.05	0.07	0.07	3.88	1.97	1556.48	-506.60	19.38
etk1a	0	8	2.46	2.12	10.16	1.28	0.18	0.04	0.03	0.04	4.00	2.17	1540.74	-459.16	18.95
et1a	0	24	1.91	1.71	11.45	1.20	0.07	0.01	0.01	0.02	4.97	2.75	792.43	-363.64	17.79
etk5b	2	0.5	1.80	1.57	7.20	0.56	2.00	0.16	0.46	0.37	4.24	1.66	1083.26	-758.25	21.20
etk4b	2	1	1.39	1.23	8.43	2.22	1.17	0.31	0.25	0.22	4.31	2.31	1004.44	-626.55	20.31
etk3b	2	2	2.04	1.79	8.97	0.81	0.62	0.06	0.13	0.13	3.96	1.91	1319.44	-523.88	19.30
etk2b	2	4	2.01	1.80	10.53	2.54	0.37	0.09	0.07	0.07	3.86	1.81	1402.78	-513.87	19.70
etk1b	2	8	1.54	1.33	11.11	3.11	0.19	0.05	0.04	0.04	4.57	1.60	1286.42	-426.48	19.03
et1b	2	24	2.35	1.98	23.13	0.38	0.13	0.01	0.04	0.04	13.97	7.42	1083.26	-130.45	14.41

Table 3. Textural parameters and growth rates estimated for titanomagnetite. L_{max}: maximum crystal length, average of 10 largest crystals. G_{max}: maximum crystal growth rate, average of 10 largest crystals. G_{batch}: crystal growth rate as determined by dividing characteristic crystal size (d) by time. G_{CSD}: crystal growth rate estimated from the slope of the crystal size distribution. d: characteristic crystal size. L_d: characteristic size determined from the slope of the crystal size distribution. N_A: area number density. See text for equations defining the parameters.

Table 1S. Representative microprobe analyses of clinopyroxenes from hydrous and anl

#	SiO ₂	TiO ₂	Al ₂ O ₃	FeO	MgO	CaO	Na ₂ O	K ₂ O	Si	Ti
et1a	44.93	1.25	10.25	9.55	12.00	19.66	0.54	0.00	1.71	0.04
et1a	44.50	1.13	9.90	9.39	12.14	19.98	0.58	0.00	1.71	0.03
et1a	45.52	1.12	9.79	9.22	12.01	20.04	0.61	0.02	1.73	0.03
et1a	44.06	1.18	10.39	9.32	11.98	20.14	0.61	0.00	1.69	0.03
et1a	45.69	1.07	9.71	8.90	12.20	20.17	0.61	0.01	1.73	0.03
et1a	44.36	1.26	10.27	9.42	11.82	20.32	0.65	0.01	1.70	0.04
et1a	44.82	1.20	10.32	9.25	11.89	20.12	0.64	0.00	1.71	0.03
et1a	45.04	1.15	10.34	9.37	11.94	19.99	0.65	0.00	1.71	0.03
et1a	44.21	1.21	10.30	9.30	11.93	19.87	0.56	0.01	1.70	0.04
et1a	45.47	1.11	9.35	8.80	12.58	20.43	0.64	0.01	1.73	0.03
et1a	46.28	1.11	9.04	8.82	12.44	20.31	0.66	0.01	1.75	0.03
et1a	44.38	1.28	10.47	9.39	11.79	20.20	0.65	0.01	1.70	0.04
et1a	45.23	1.17	9.97	9.49	12.11	19.88	0.60	0.01	1.72	0.03
et1a	45.33	1.25	10.46	9.33	11.60	20.08	0.62	0.00	1.72	0.04
et1a	44.57	1.14	10.45	9.27	11.75	20.06	0.64	0.00	1.71	0.03
et1a	45.39	1.15	9.91	9.50	12.03	20.02	0.63	0.01	1.72	0.03
et1a	45.98	1.08	9.05	8.90	12.59	20.13	0.64	0.01	1.75	0.03
et1a	44.49	1.21	10.38	9.45	11.87	19.94	0.58	0.00	1.70	0.03
et1a	44.87	1.35	10.60	9.70	11.55	20.06	0.59	0.00	1.70	0.04
et1a	46.18	1.14	9.55	9.11	12.13	20.12	0.66	0.01	1.74	0.03
etk1a	44.38	1.29	10.09	9.95	12.48	19.51	0.65	0.01	1.70	0.04
etk1a	44.36	1.19	10.01	9.98	12.53	19.51	0.64	0.01	1.70	0.03
etk1a	45.00	1.19	9.87	9.63	12.58	19.79	0.68	0.03	1.71	0.03
etk1a	45.00	1.15	9.81	9.73	12.57	19.41	0.64	0.03	1.72	0.03
etk1a	44.16	1.37	10.35	9.83	12.06	19.42	0.62	0.02	1.69	0.04
etk1a	44.21	1.40	10.67	9.75	12.11	19.95	0.70	0.01	1.68	0.04
etk1a	44.26	1.28	10.41	9.59	12.09	19.86	0.65	0.01	1.69	0.04
etk1a	43.99	1.24	10.48	9.78	12.09	19.96	0.62	0.02	1.69	0.04
etk1a	44.15	1.30	10.39	9.66	12.13	20.12	0.66	0.02	1.69	0.04
etk1a	44.36	1.29	10.39	9.64	12.07	20.02	0.61	0.01	1.69	0.04
etk1a	44.14	1.27	10.57	9.78	12.13	19.80	0.67	0.03	1.69	0.04
etk1a	45.29	1.23	9.69	9.47	12.57	19.80	0.64	0.01	1.72	0.04
etk1a	44.73	1.33	10.04	9.68	12.41	19.88	0.62	0.01	1.70	0.04
etk1a	44.82	1.22	10.12	9.66	12.33	19.75	0.65	0.03	1.71	0.03
etk1a	44.14	1.31	10.58	9.87	11.99	19.78	0.64	0.02	1.69	0.04
etk1a	44.42	1.25	10.36	9.64	12.17	19.85	0.62	0.02	1.70	0.04
etk1a	44.67	1.27	10.22	9.46	12.27	20.03	0.71	0.01	1.70	0.04
etk1a	44.42	1.24	10.33	9.46	12.33	19.90	0.67	0.02	1.69	0.04
etk1a	44.81	1.17	10.40	9.58	12.32	19.87	0.71	0.01	1.70	0.03
etk1a	45.95	1.15	9.20	8.97	13.05	19.84	0.79	0.02	1.74	0.03